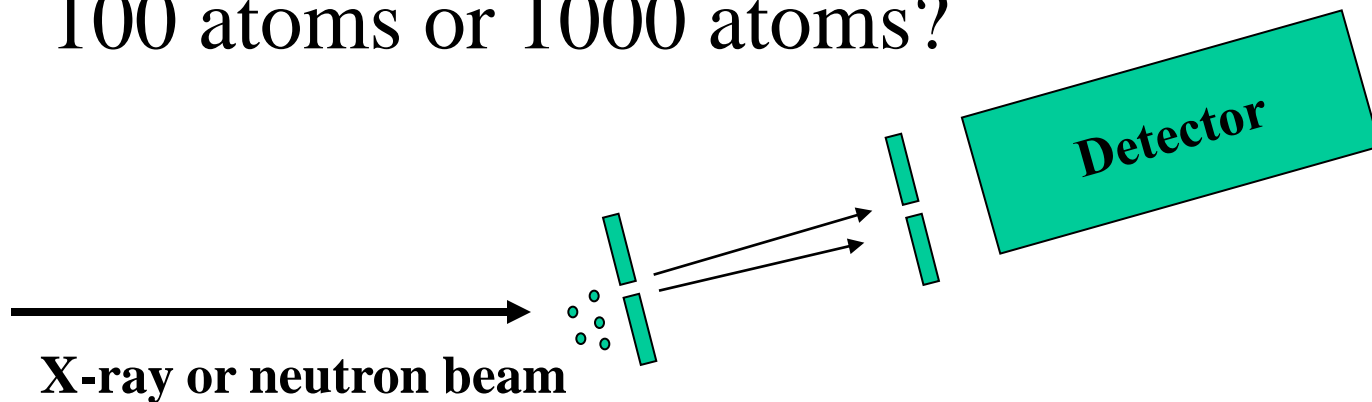


Diffuse Scattering

- Anticipatory (trick) question: If you have an x-ray or neutron detector looking at a small sample volume, which will scatter more x-rays or neutrons into the detector 1 atom 100 atoms or 1000 atoms?



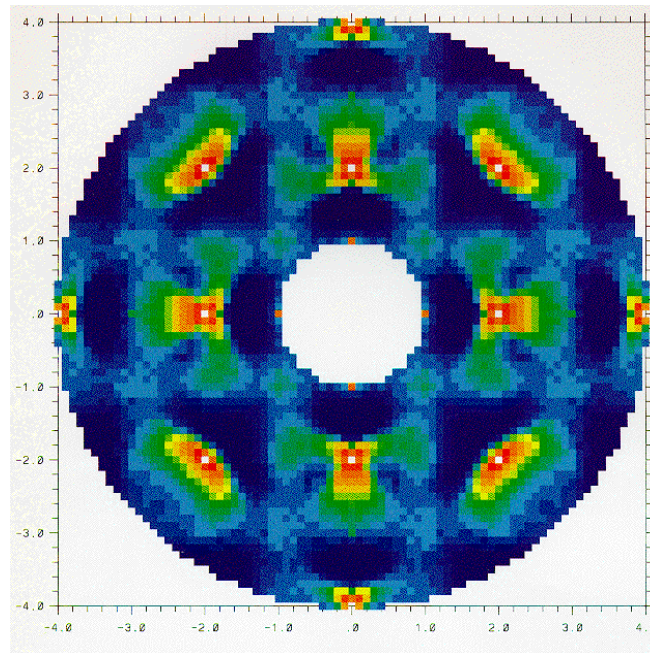
Answer: Depends!

Diffuse Scattering

Gene E. Ice

Materials Science and Technology Division

Oak Ridge National Laboratory, USA



National School on Neutron and X-ray Scattering
ORNL/SNS June 2010

Presentation concentrates year graduate-level course into 1 hour

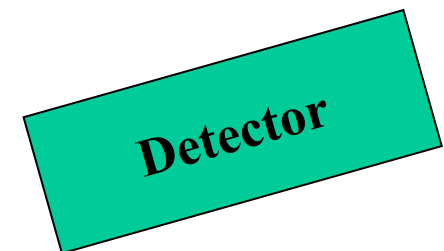
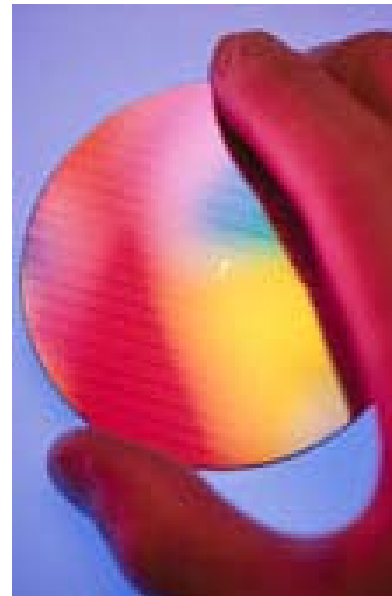
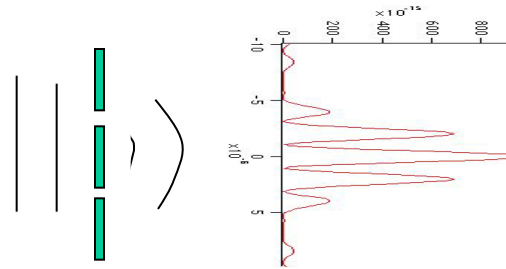
- Skip mathematical complexities
- Expose to range of applications
- Develop *intuition* for length scales
- Talk like x-ray/neutron scattering guru
 - *Reciprocal space*
 - *Debye Temperature*
 - *Laue monotonic*
 - *Krivoglaz defects of 1st/2nd kinds!*



Great for cocktail parties or impressing attractive strangers

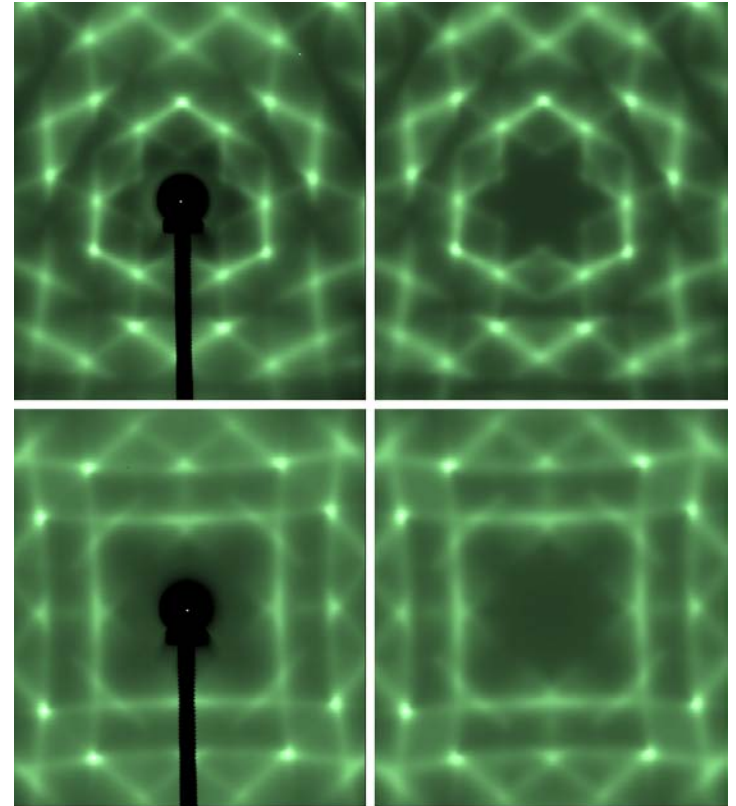
Distribution of atoms redistributes scattering

- **Familiar light example**
- **Practical applications- zero background plates for powder diffraction**
- **Some problems- finding reflections**
- **Wave→diffraction**



Diffuse scattering poised for a revolution!

- Synchrotron sources /new tools enable new applications
 - Rapid measurements for combinatorial and dynamics
 - High energy for simplified data analysis
 - Small (dangerous) samples
- Advanced neutron instruments emerging
 - Low Z elements
 - Magnetic scattering
 - Different contrast
- New theories provide direct link between experiments and first-principles calculations



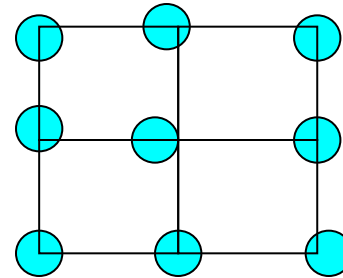
Experiment

Theory

Major controversies have split leading scientists in once staid community!

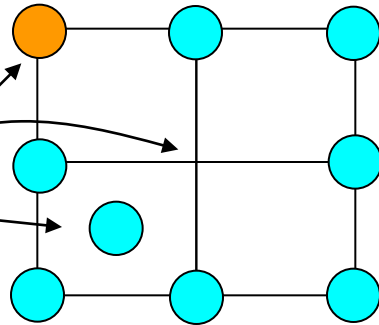
Diffuse scattering due to *local* (short ranged) correlations/ fluctuations

- Thermal diffuse scattering (TDS) →

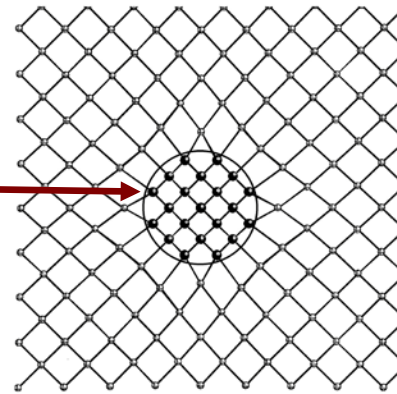


- Point defect

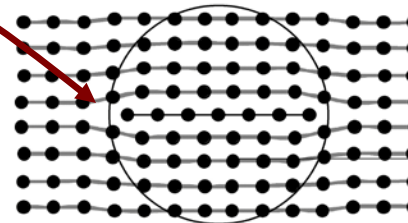
- Site substitution
- Vacancy
- Interstitial



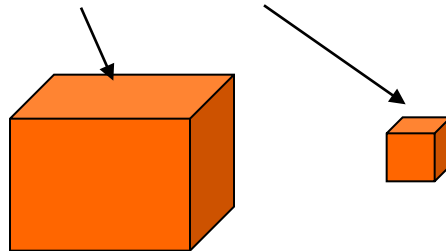
- Precipitate



- Dislocations



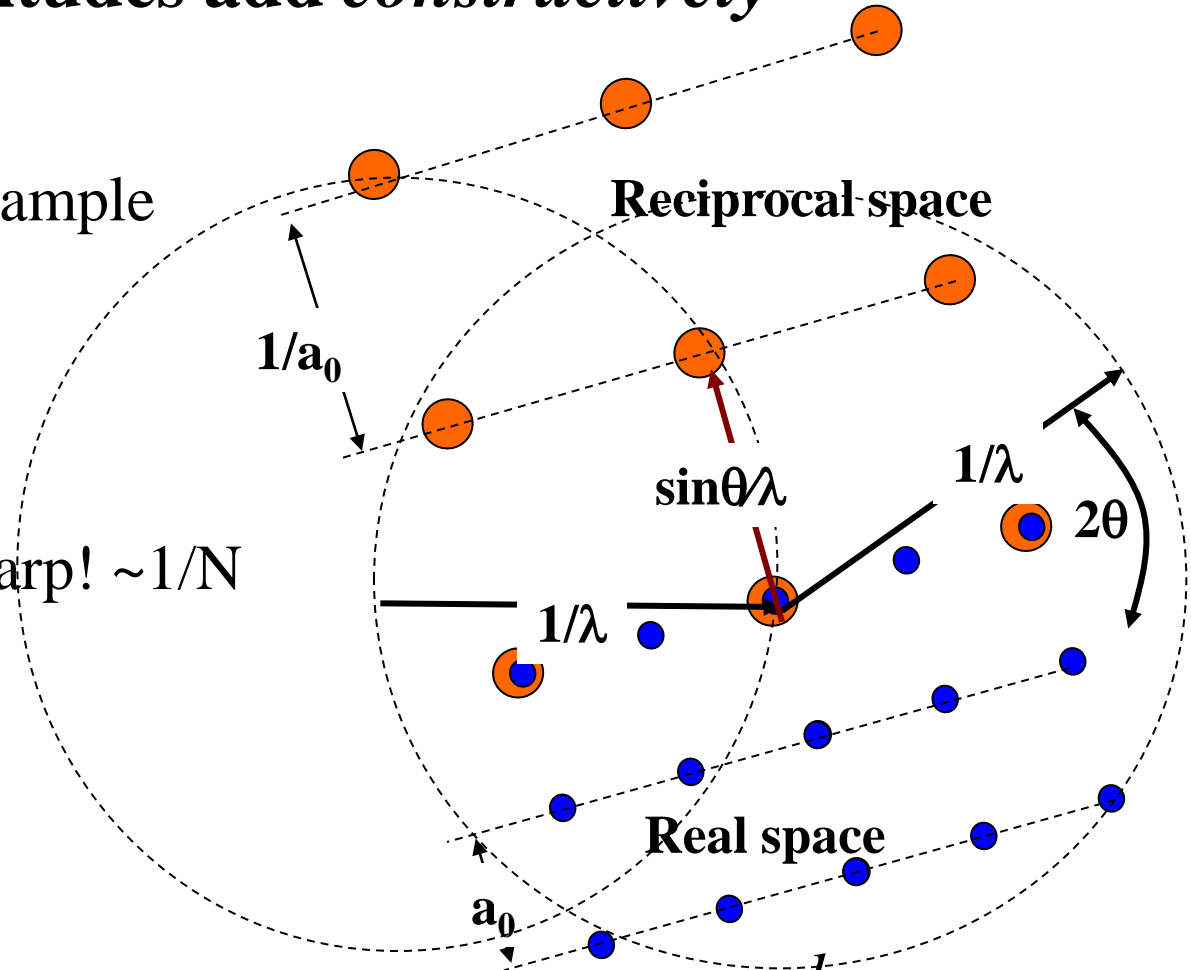
- Truncated surface- more



All have in common reduced correlation length!

Bragg reflection occurs when scattering amplitudes add *constructively*

- Orientation of sample
- Wavelength
- Bragg Peaks sharp! $\sim 1/N$ (arc seconds)



Think in terms of *momentum transfer* $\vec{p}_0 = \frac{h}{\lambda} \hat{d}_0$

Length scales

- Big real \rightarrow small reciprocal
- Small real \rightarrow big reciprocal
- *We will see same effect in correlation length scales*
 - Long real-space correlation lengths scattering close to Bragg peaks


$$2\sin\theta/\lambda$$

2θ

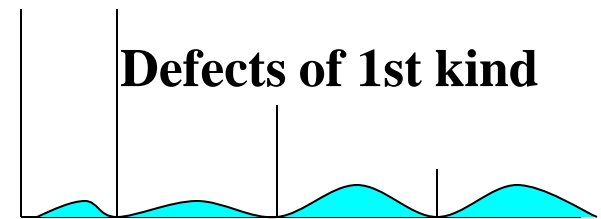
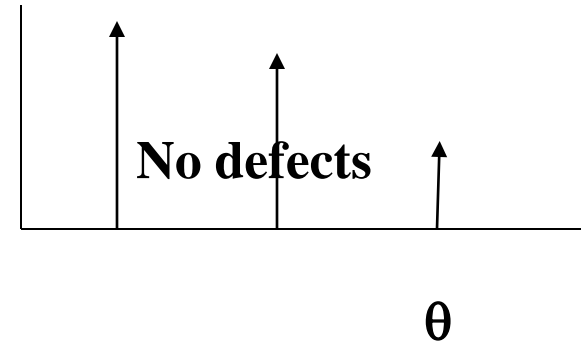
If you remember nothing else!

Krivoglaz classified defects by effect on Bragg Peak

- **Defects of 1st kind**

- Bragg width unchanged
- Bragg intensity decreased
- Diffuse redistributed in reciprocal space
- Displacements remain finite

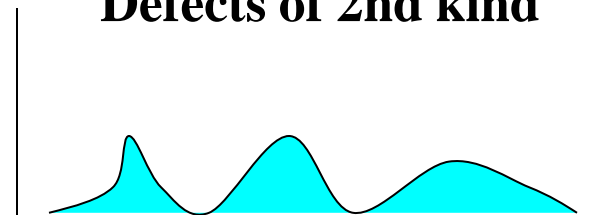
Intensity



- **Defects of 2nd kind**

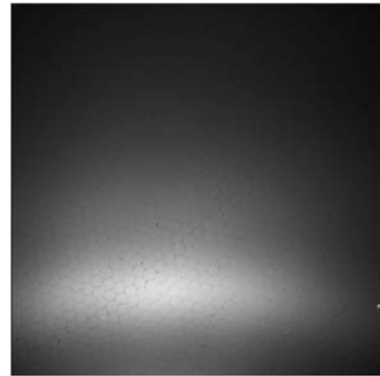
- No longer distinct Bragg peaks
- Displacements continue to grow with crystal size

Defects of 2nd kind



Dimensionality influences diffraction

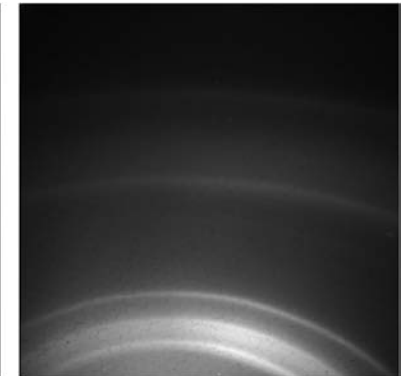
- Small size → broad diffraction



a. Amorphous



b. nanocrystalline

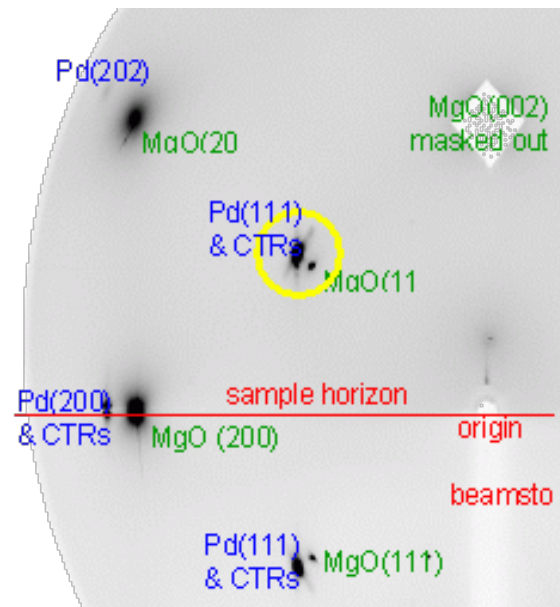


c. crystalline

- Polycrystalline

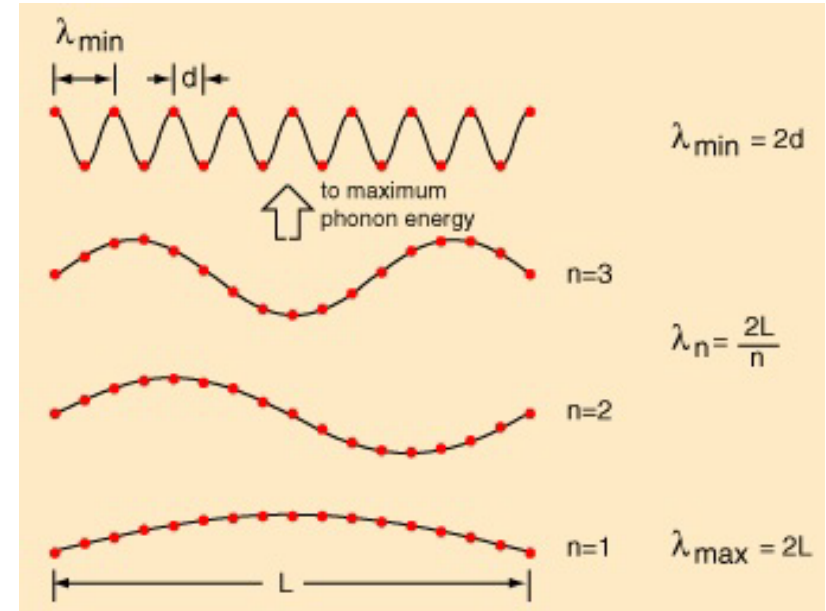
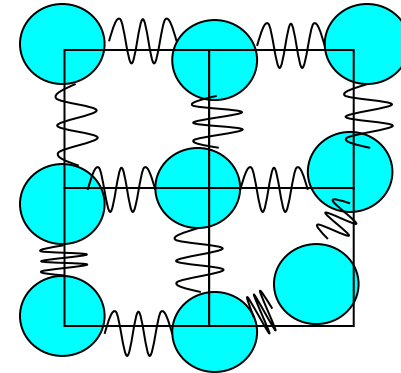


- Single crystalline (roughness)



Thermal motion-Temperature Diffuse Scattering-(TDS) -defect of 1st kind

- Atoms coupled through atomic bonding
- Uncorrelated displacements at distant sites
 - (finite)
- Phonons (wave description)
 - Amplitude
 - Period
 - Propagation direction
 - Polarization (transverse/compressional)



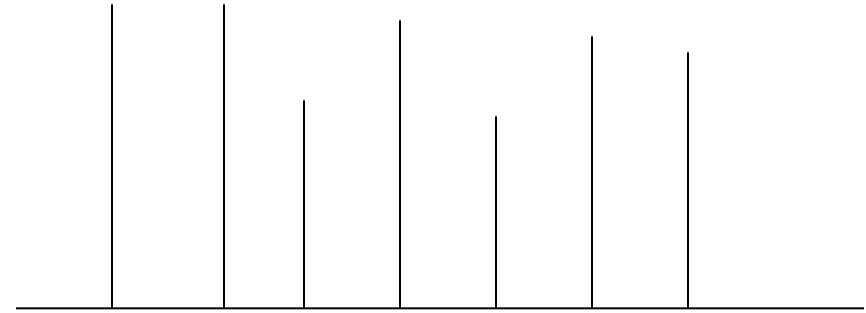
**Sophisticated theories from
James, Born Von Karmen, Krivoglaz**

A little math helps for party conversation

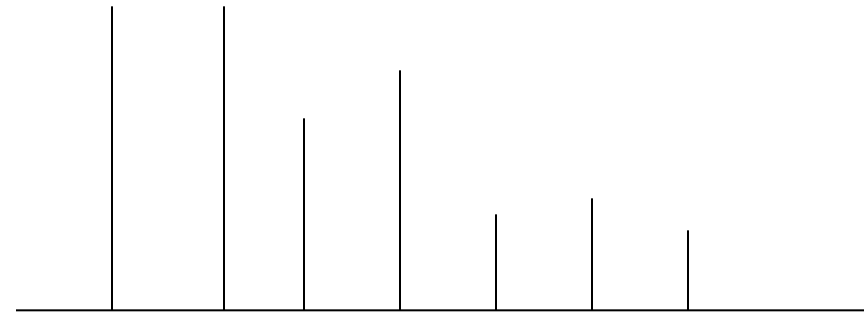
- Decrease in Bragg intensity scales like e^{-2M} , where

$$2M = 16\pi^2 \left\langle u_s^2 \right\rangle \frac{\sin^2 \theta}{\lambda^2}$$

- Small* $\theta \rightarrow$ *Big* reflections
- e^{-2M} shrinks (*bigger* effect) with θ (q)



Low Temperature

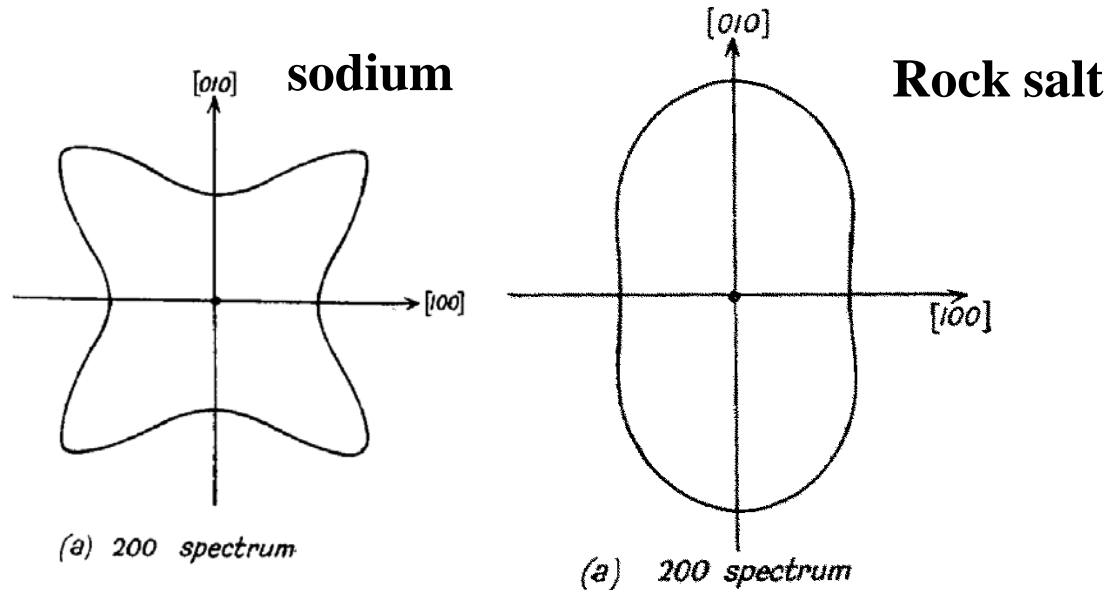


High temperature

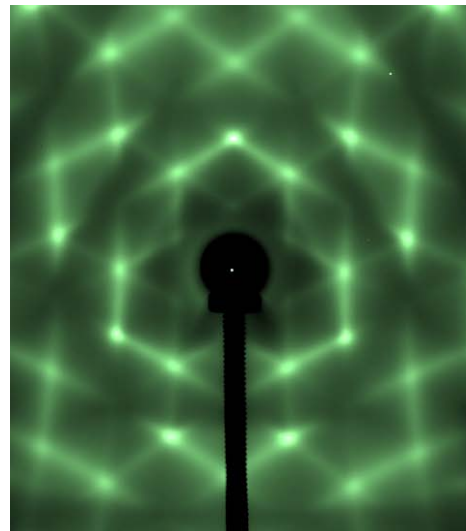
Displacements, u_s depend on *Debye Temperature* θ_D - *Bigger* θ_D
 \rightarrow *smaller* displacements !

TDS makes beautiful patterns reciprocal space

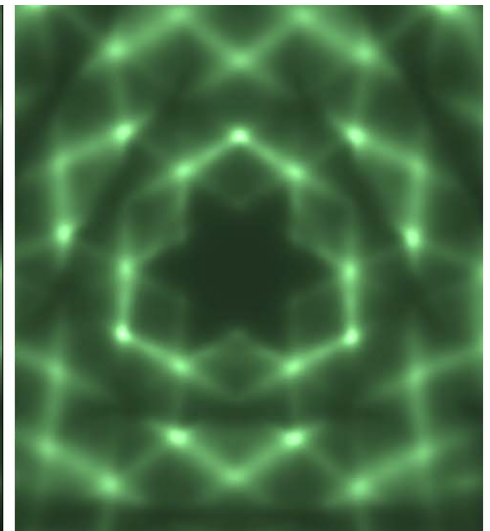
- Iso- intensity contours
 - Butterfly
 - Ovoid
 - Star
- Transmission images reflect symmetry of reciprocal space and TDS patterns



Experiment



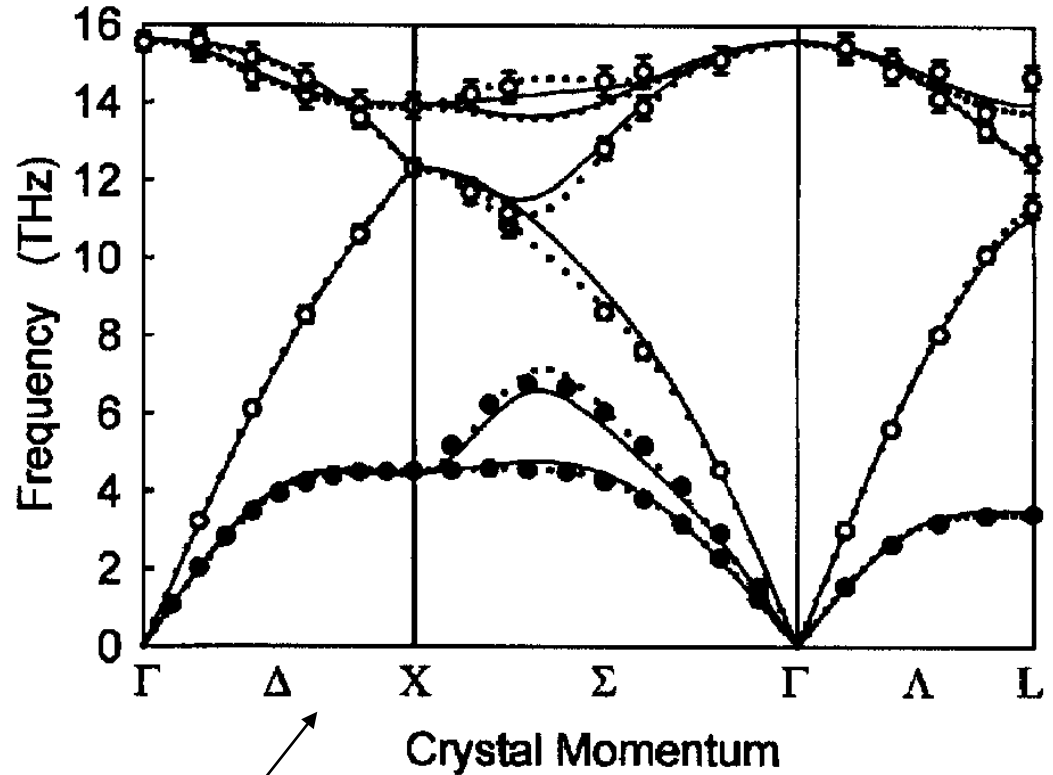
Theory



**Chiang et al. Phys. Rev. Lett.
83 3317 (1999)**

***X-rays scattering measurements infer* phonon dispersion from quasi-elastic scattering**

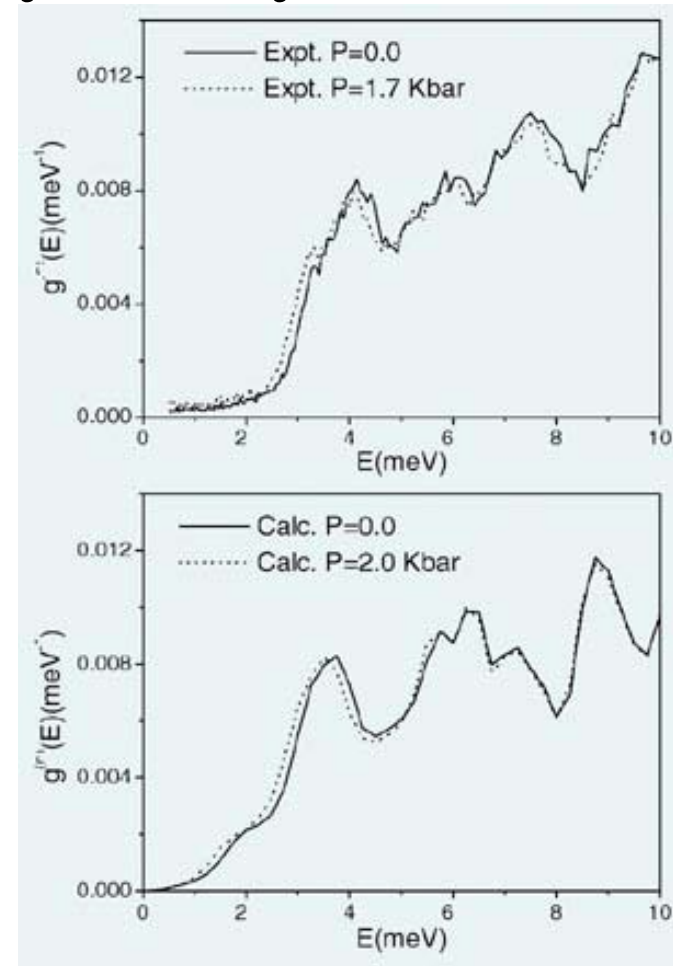
- Phonon energies *milli-eV*
- Synchrotron based high-E resolution X-ray beamlines can measure phonons *in some cases*
- Emerging area for high-brilliance x-ray sources



Phonon spectrum gives natural vibration frequencies in different crystal directions!

Inelastic neutron/x-ray scattering directly measures phonon spectra in symmetry directions

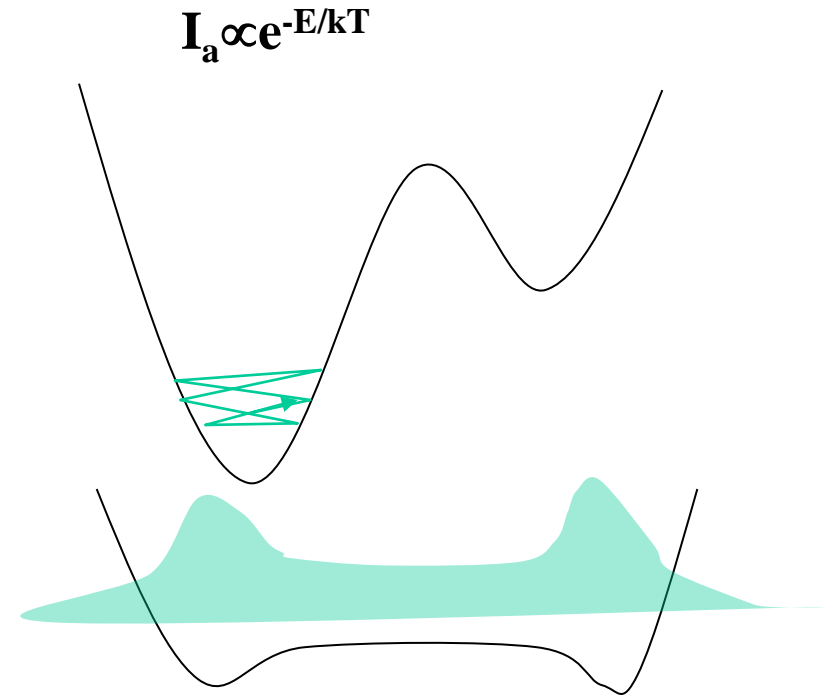
- Inelastic neutron scattering confirms origins of negative Grüneisen coefficient in cubic ZrW_2O_8 (negative thermal expansion)-disordering phase transition.
- Unusual thermal displacements *often associated with phase transitions*.

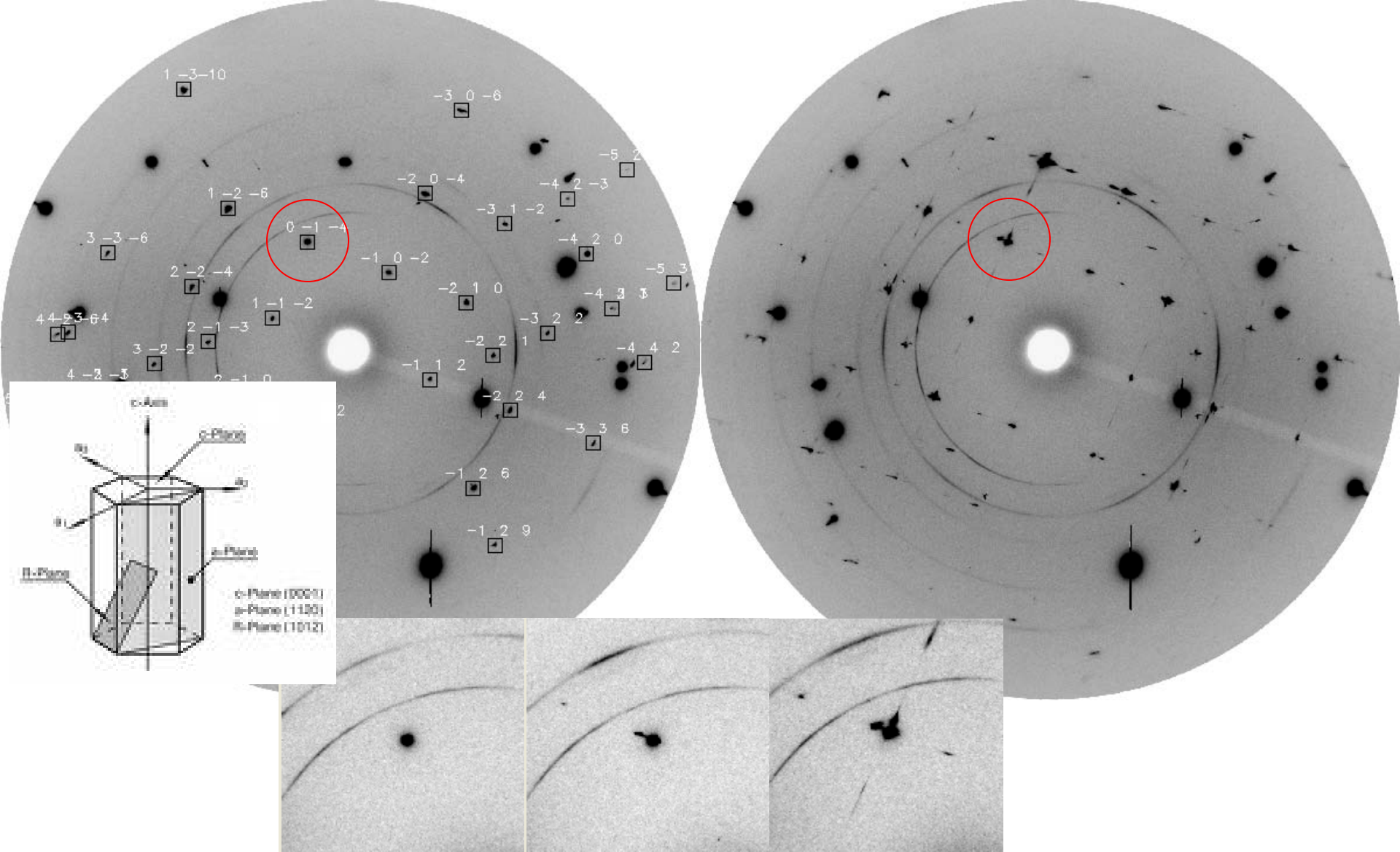


Phonon energies similar to meV neutron energies.

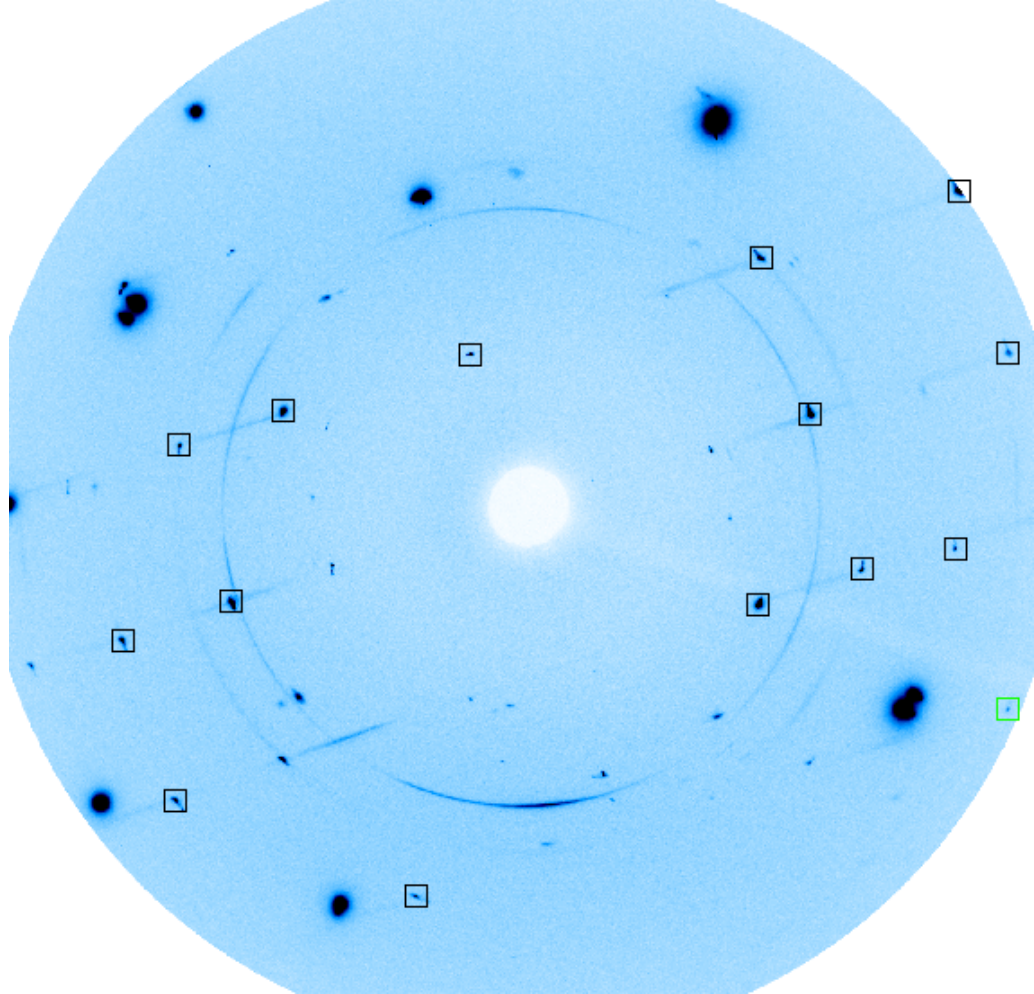
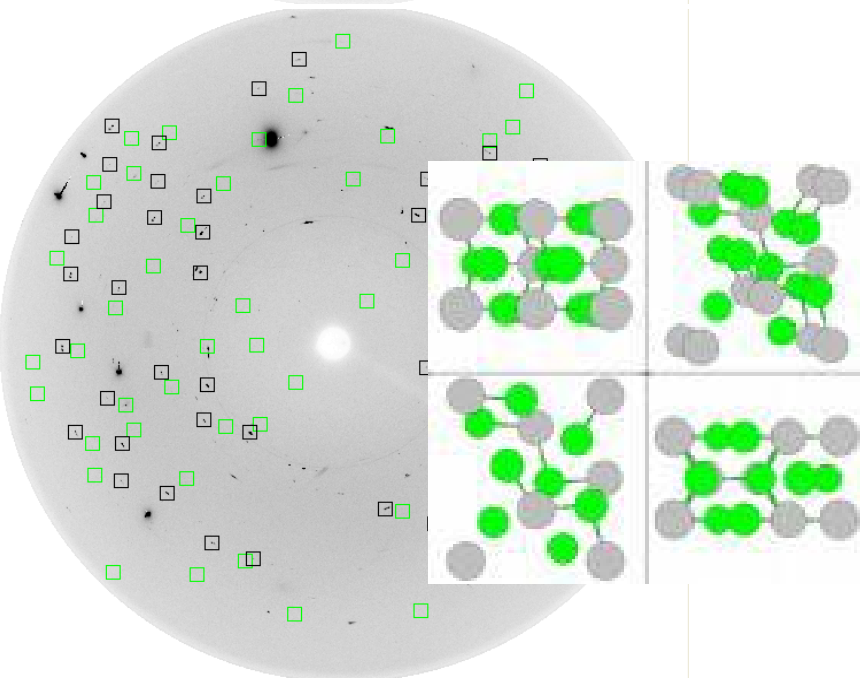
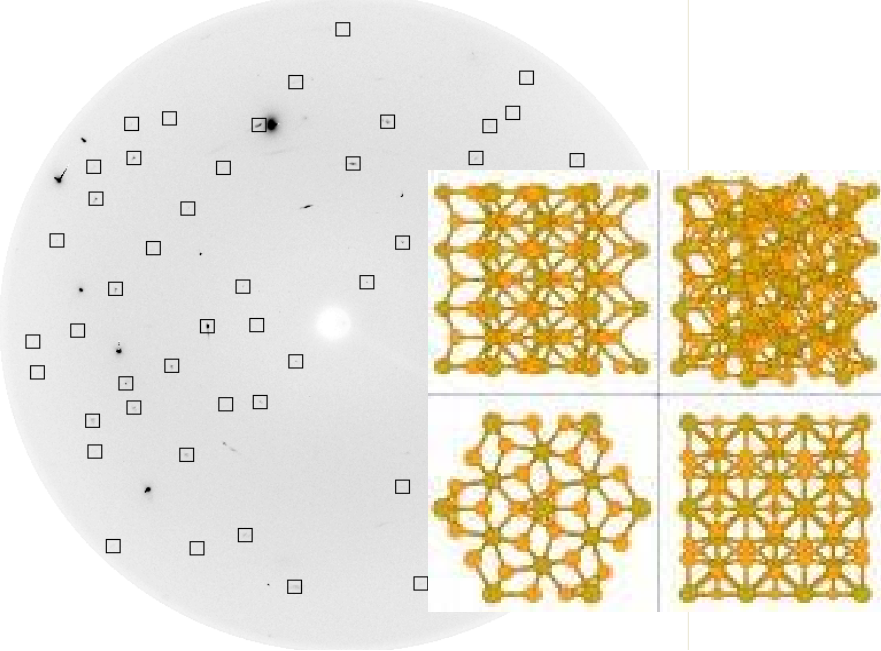
Diffuse scattering near phase transitions

- **Distribution of configurations at finite temperature**
 - Mixed phases (1st order)
- **Extended displacements**
- **High-pressure**
 - higher-co-ordination
 - Longer NN bond distance
 - Smaller volume/atom





R-3c \rightarrow I2/a displacive transition observed in a single crystal of Cr_2O_3 at 80 GPa



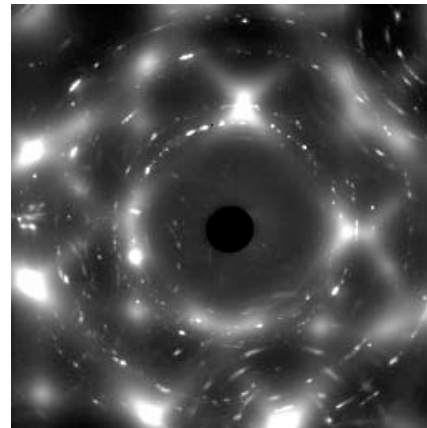
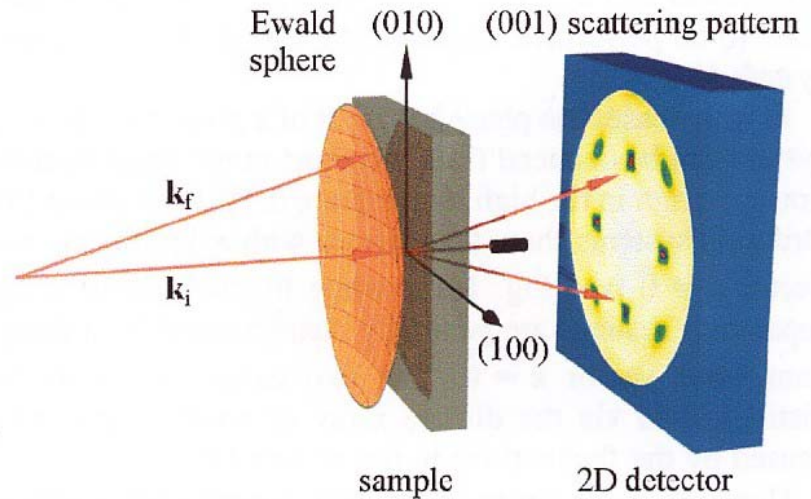
Diffuse scattering before the transformation occurs, heating at ~1000 K

C22→C23 transition in Fe₂P at 10 GPa
Dera et al. (2008) *Gophys. Res. Lett.*, **35**, L10301

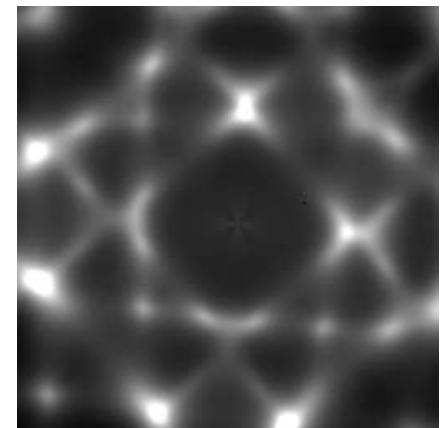
Complete transformation induced by heating the sample to 2000 K

High-energy Synchrotron X-rays are revolutionizing TDS measurements

- Small samples
- Fast (time resolved/combinatorial)
 - Experiments in seconds rather than days
- Materials that cannot be studied with neutrons



Pu experiment

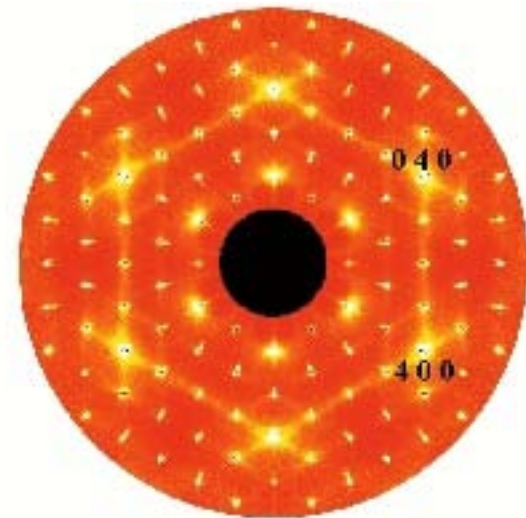
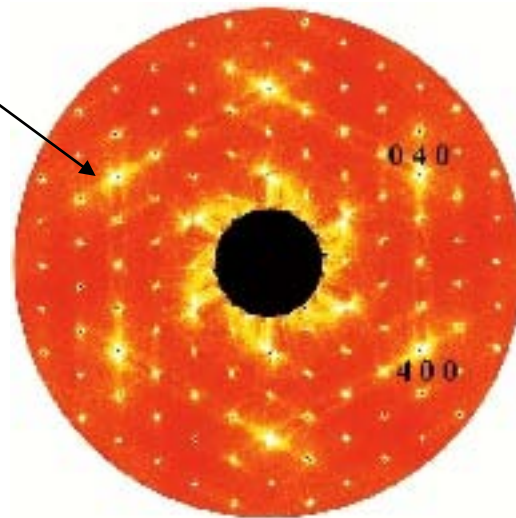


Pu theory

Neutrons uniquely sensitive to low Z

- Deuterium cross section large
- Phonon energy comparable to neutron energy
- New instruments will accelerate data collection

splitting



Welberry et al. ISIS

Experiment

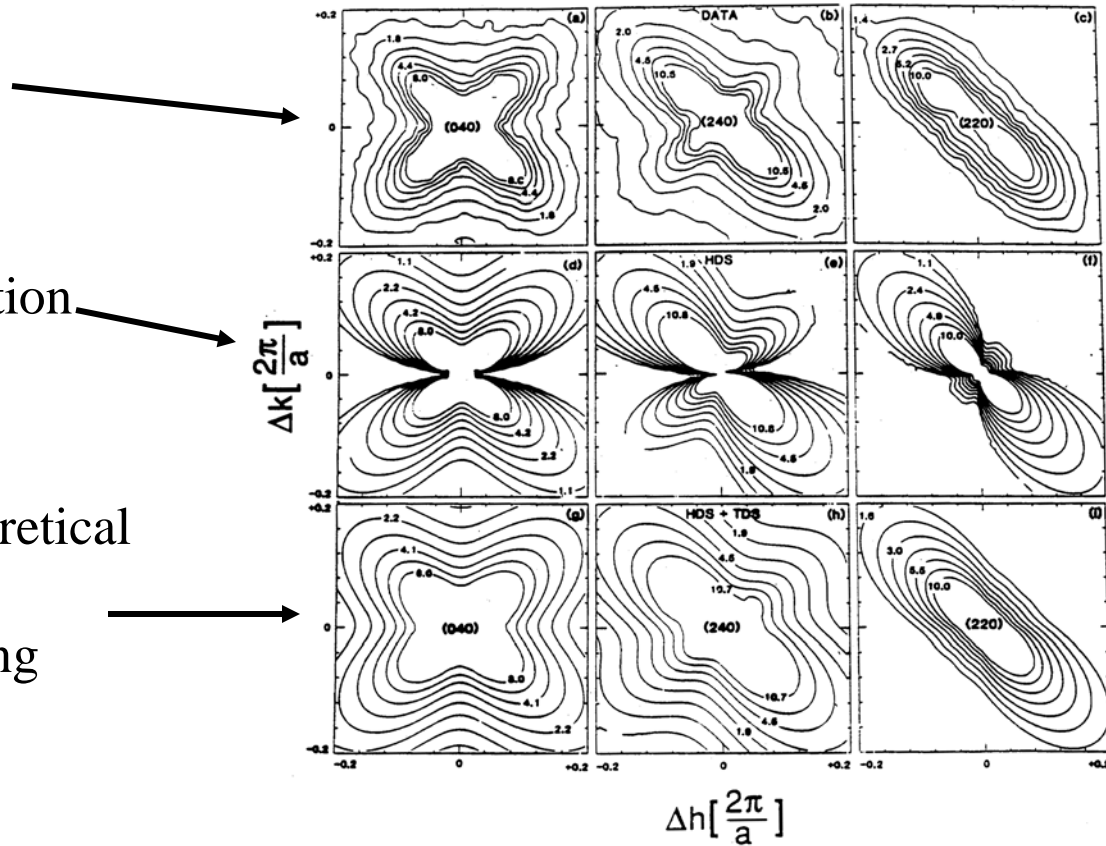
Theory

Often TDS mixed with additional diffuse scattering

- Experiment

- Strain contribution

- Combined theoretical TDS and strain diffuse scattering

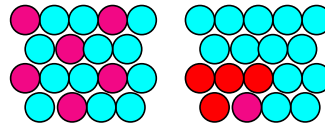


TDS must often be removed to reveal other diffuse scattering

Alloys can have another *type 1* defect-*site substitution*

Each Au has 8 Cu near-neighbors

- Long range
 - Ordering (unlike neighbors)
 - Phase separation (like neighbors)



Cu_3Au
 $L1_2$

- Short ranged
 - Ordering
 - Clustering (like neighbors)

CuAu
 $L1_0$

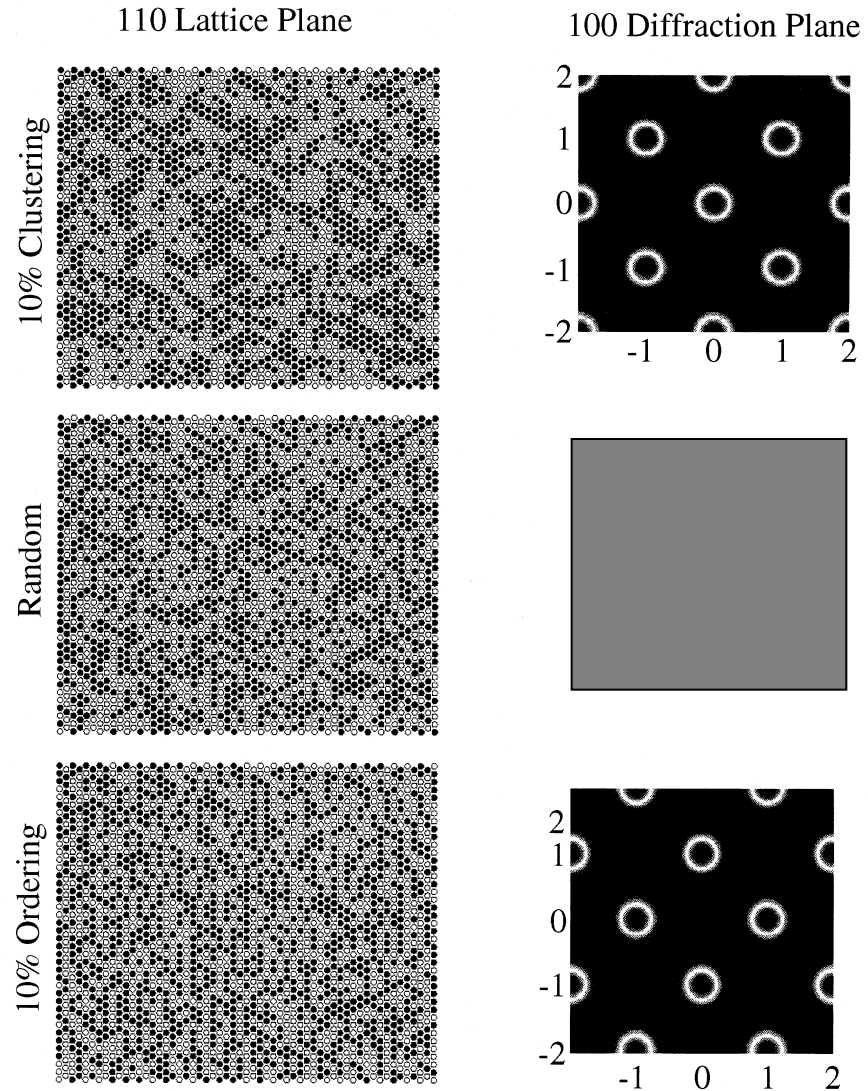
Alternating planes of Au and Cu

Redistribution depends on kind of correlation

Clustering intensity
→ fundamental sites

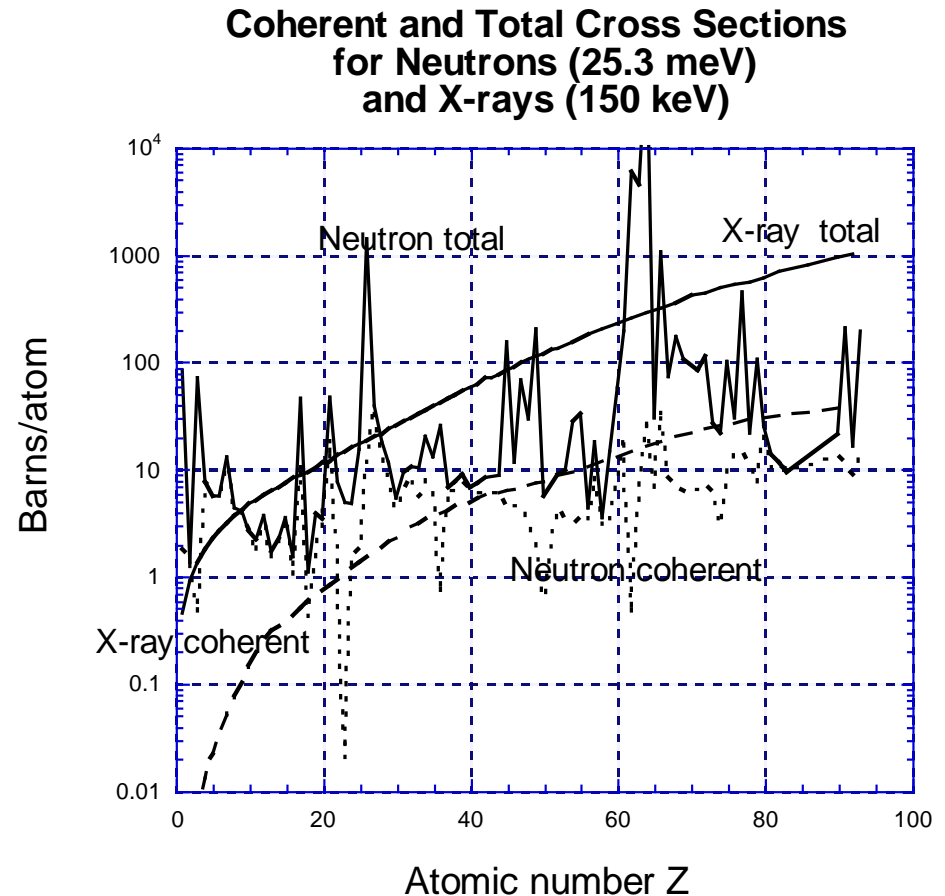
Random causes
Laue monotonic

Short-range ordering
→ superstructure sites



Neutron/ X-rays Complimentary For Short-range Order Measurements

- Chemical order diffuse scattering **proportional to contrast $(f_A - f_B)^2$**
- Neutron scattering cross sections
 - Vary wildly with isotope
 - Can have + and - sign
 - Null matrix
 - Low Z , high Z comparable
- X-ray scattering cross section
 - Monotonic like Z^2
 - Alter by anomalous scattering

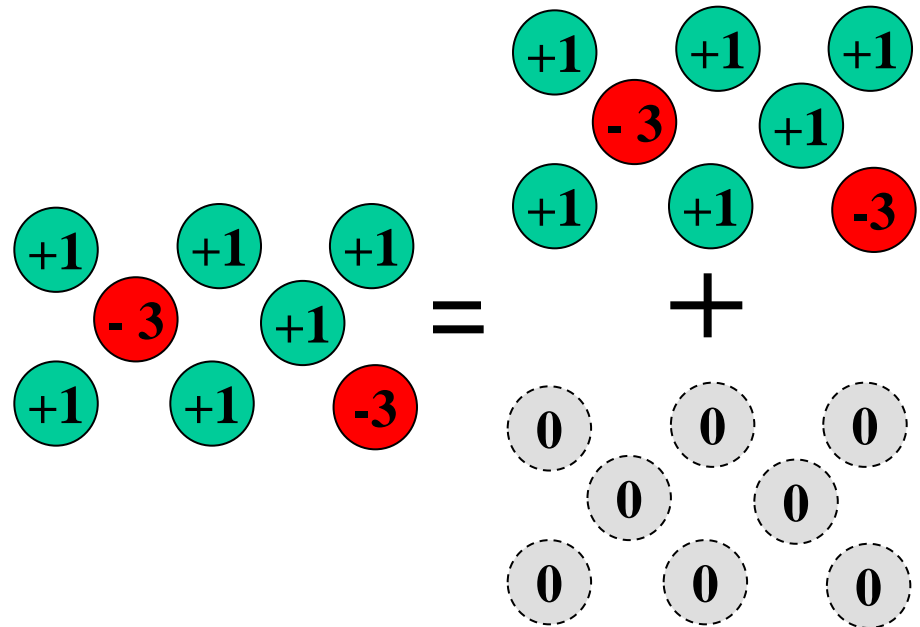


Neutrons can select isotope to eliminate Bragg scattering

- Total scattering $c_a f_a^2 + c_b f_b^2 = 3$

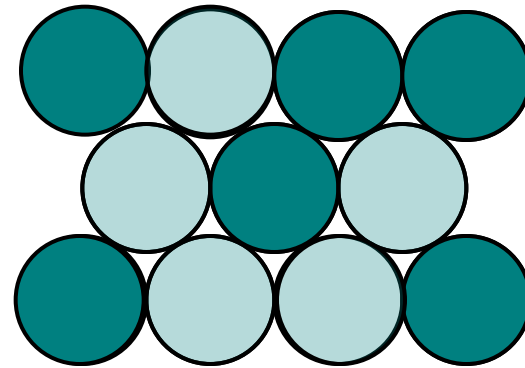
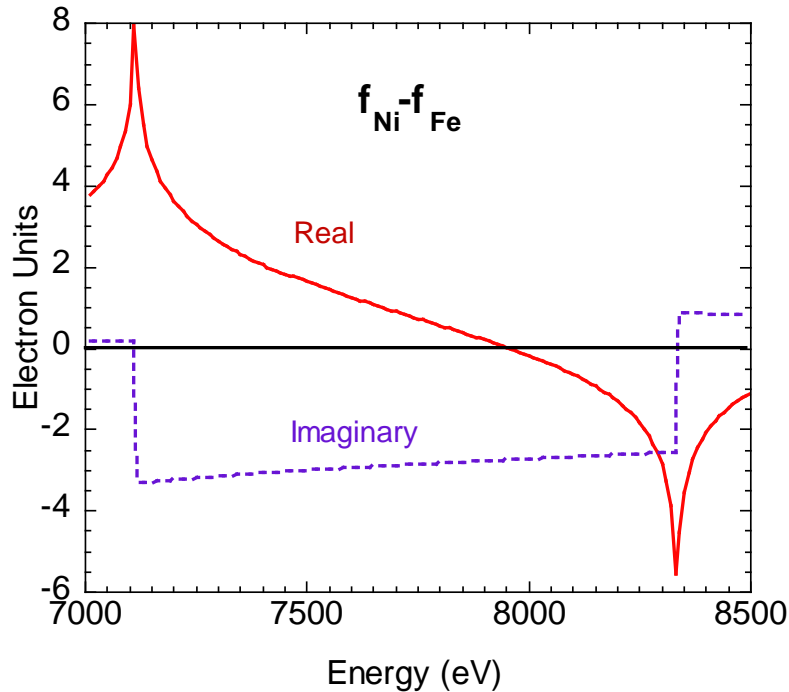
- Bragg scattering $(c_a f_a + c_b f_b)^2 = 0$

- Laue (diffuse) scattering $c_a c_b (f_a - f_b)^2 = 3$

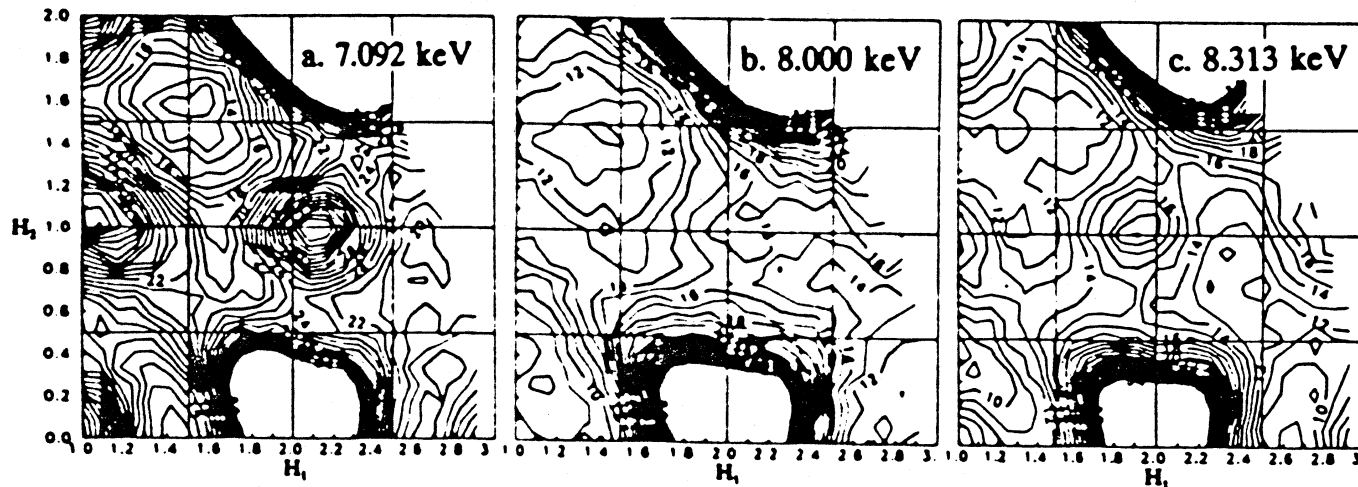


Isotopic purity important as different isotopes have distinct scattering cross sections- only one experiment ever done!

X-ray anomalous scattering can change x-ray contrast

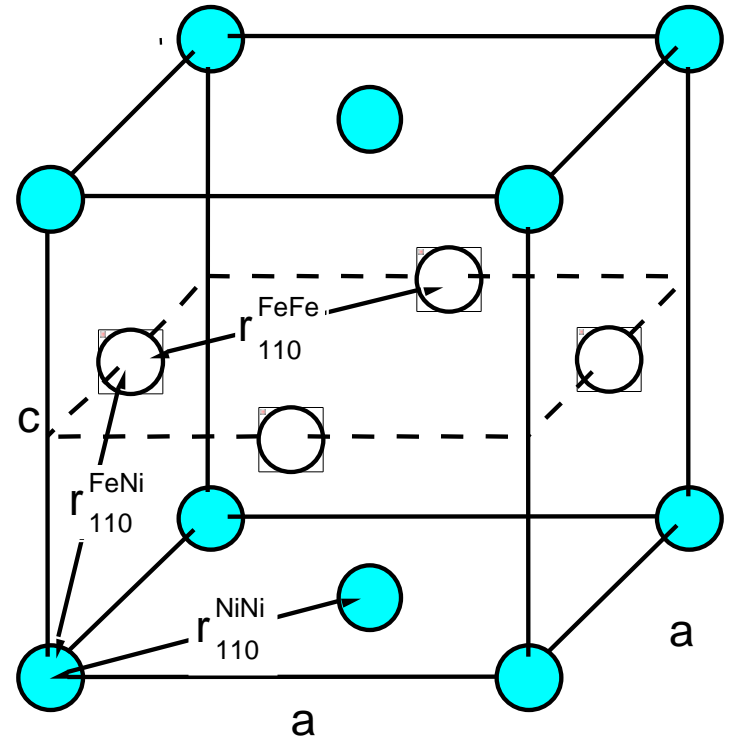


- Chemical SRO scattering scales like $(f_a - f_b)^2$
- Static displacements scale like $(f_a - f_b)$
- TDS scales like $\sim f_{\text{average}}^2$



Atomic size (static displacements) affect phase stability/properties

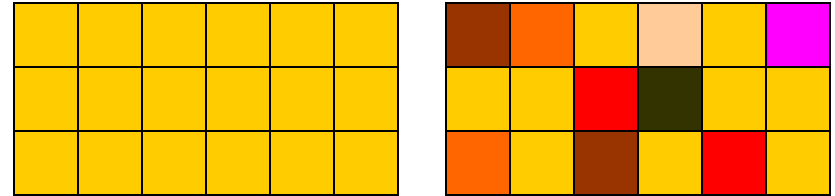
- *Ionic materials (Goldschmidt)*
 - Ratio of Components
 - **Ratio of radii**
 - Influence of polarization
- *Metals and alloy phases (hume-Rothery)*
 - **Ratio of radii**
 - Valence electron concentration
 - Electrochemical factor



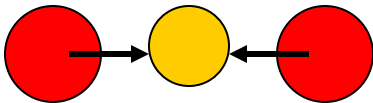
Grand challenge -include deviations from lattice in modeling of alloys

Measurement *and* theory of atomic size are hard!

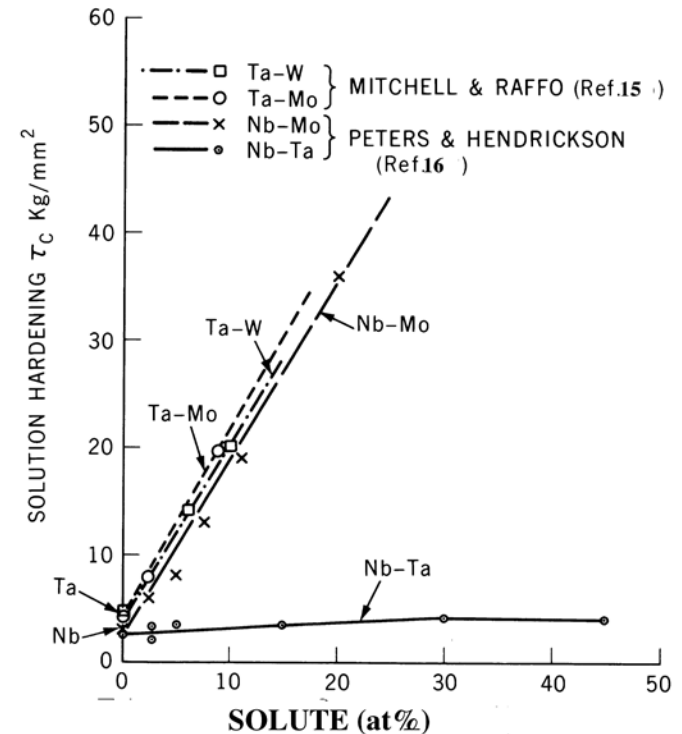
- Theory- violates repeat lattice approximation- every unit cell different!



- Experiment
 - EXAFS marginal (0.02 nm) in dilute samples
 - Long-ranged samples have balanced forces



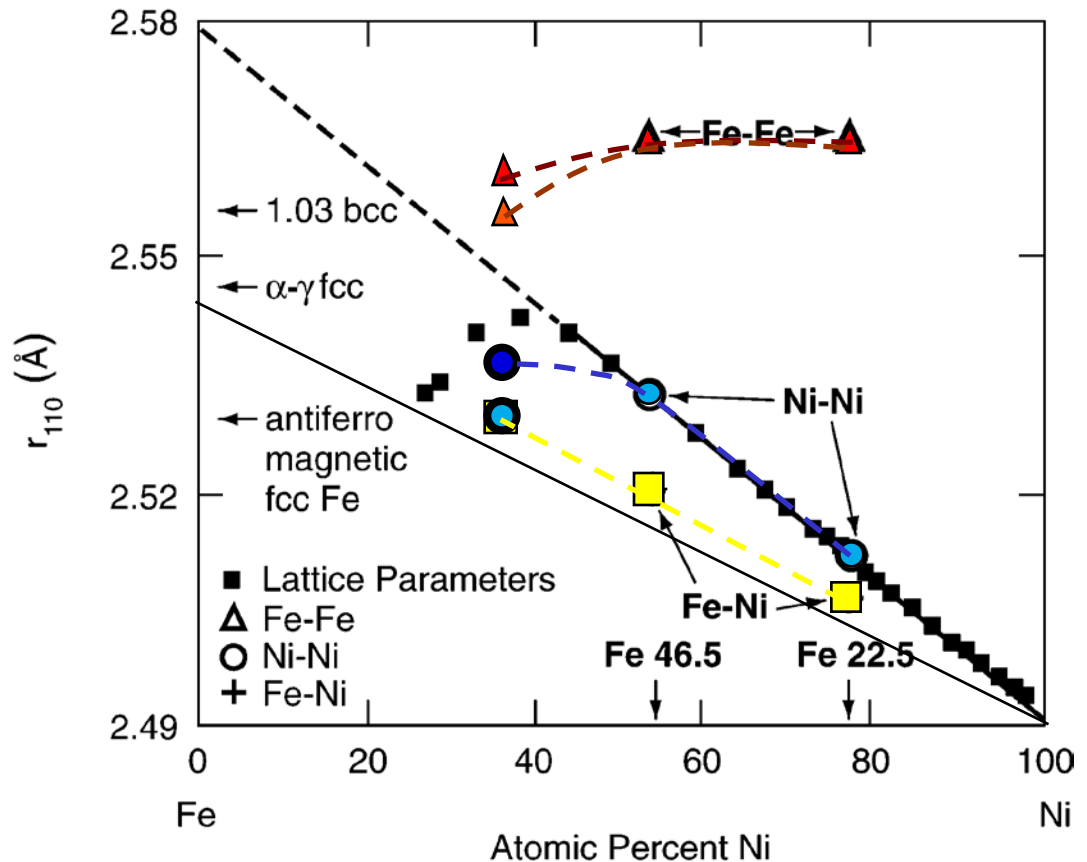
Important!



Systematic study of bond distances in Fe-Ni alloys raises interesting questions

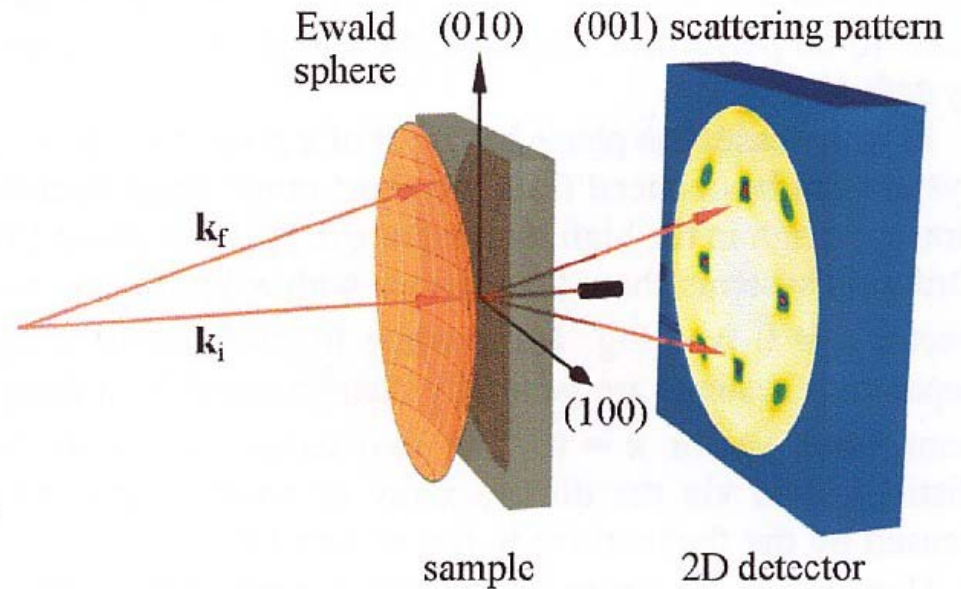
ORNL 98-7348A/rra

- Why is the Fe-Fe bond distance stable?
- Why does Ni-Ni bond swell with Fe concentration?
- Are second near neighbor bond distances determined by first neighbor bonding?

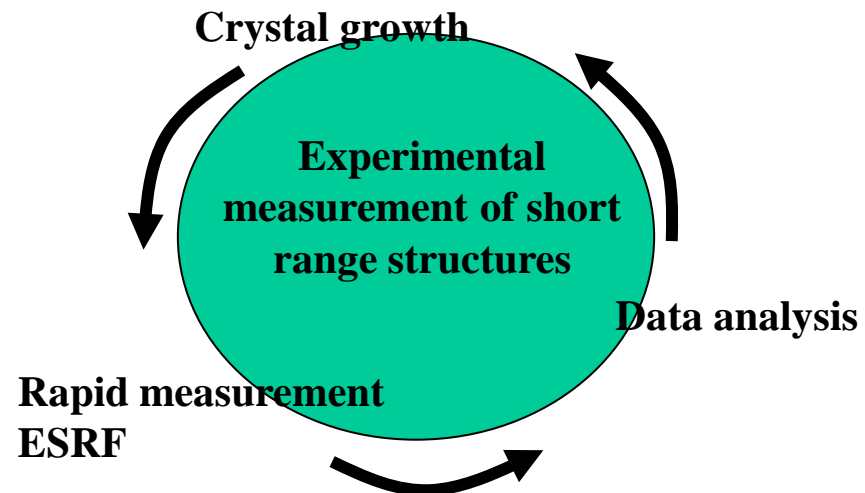


High-energy x-ray measurements revolutionize studies of phase stability

- Data in *seconds* instead of *days*
- Minimum absorption and stability corrections
- New analysis provides direct link to first-principle



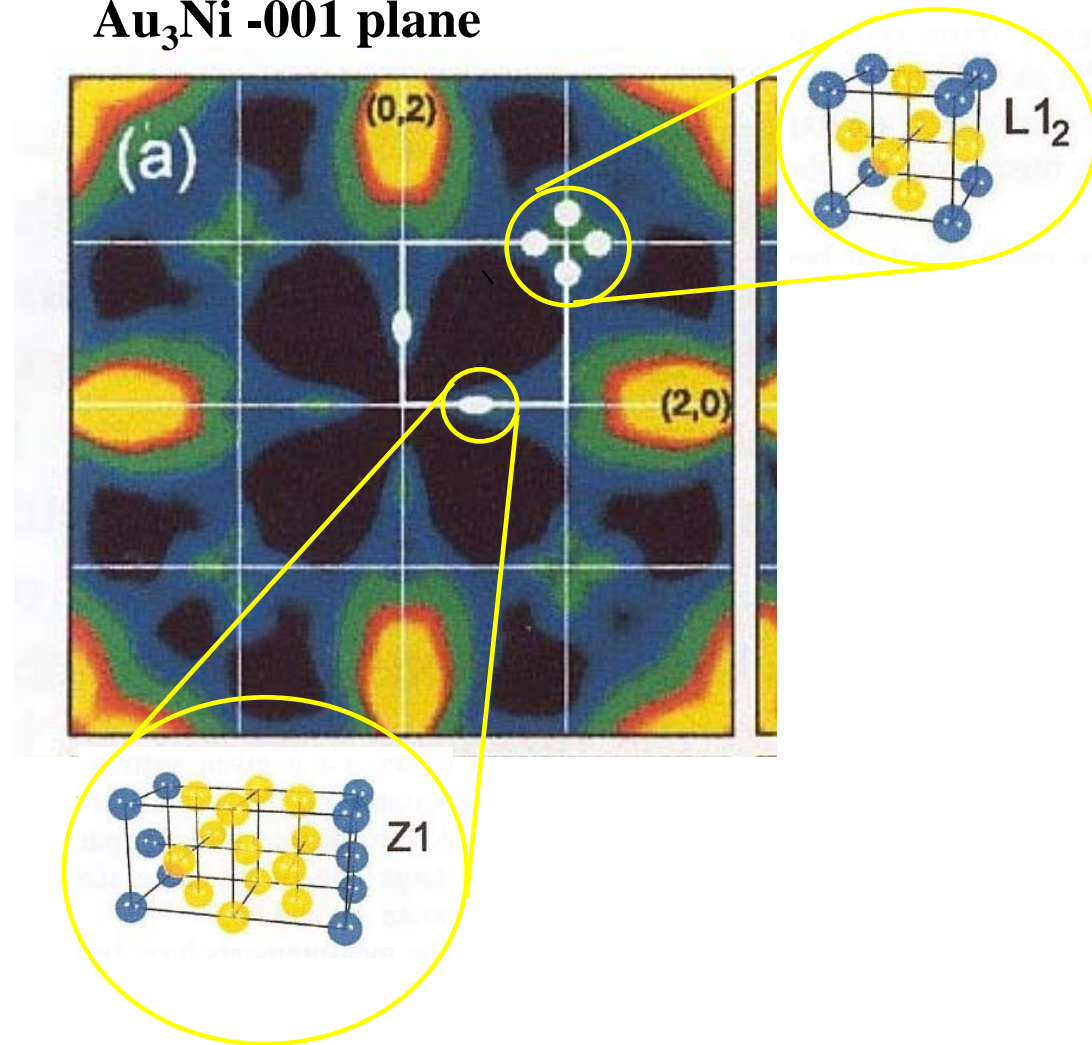
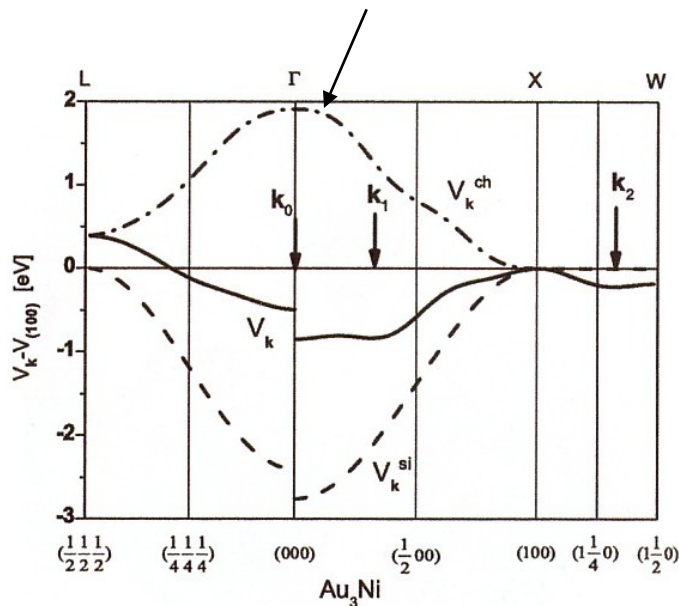
Max Planck integrates diffuse x-ray scattering elements!



Measurements show competing tendencies to order

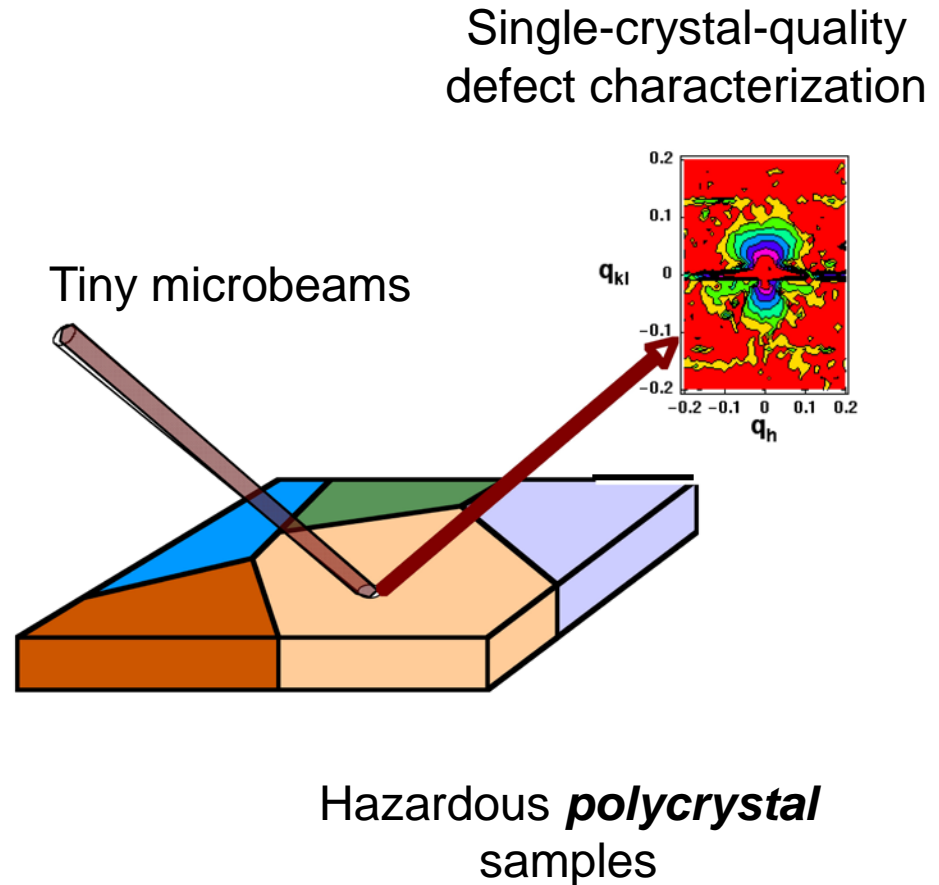
Au_3Ni -001 plane

- Both $L1_2$ and Z_1 present
- Compare with first principles calculations



Intense microbeams/area detectors provide new direction in diffuse scattering

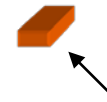
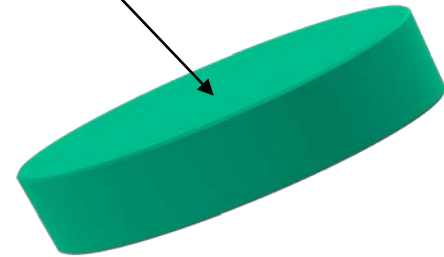
- Tiny crystals ($20\text{ }\mu\text{m}$)
 - Natural polycrystals
 - No special sample prep
- Combinatorial
- Dangerous samples



Small irradiated volumes simplify handling/preparation

- Activity \sim volume (10^{-5})
- Much less waste (10^{-7})
- Polycrystalline samples easier obtain-
closer to real materials

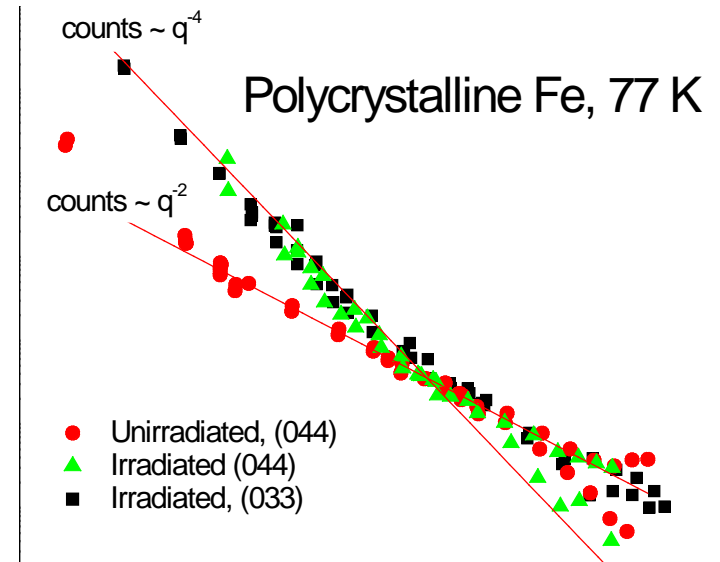
Traditional diffuse sample $\sim 300 \text{ mm}^3$



Microsample $\sim 10^{-3} \text{ mm}^3$
100-1000 samples

Diffuse microdiffraction holds promise for irradiated materials

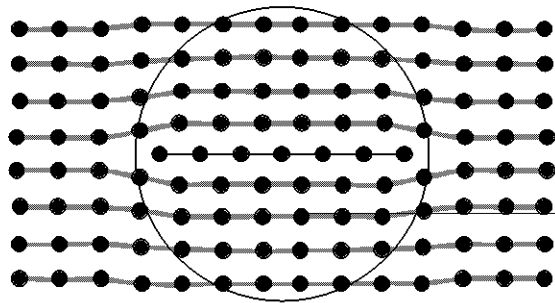
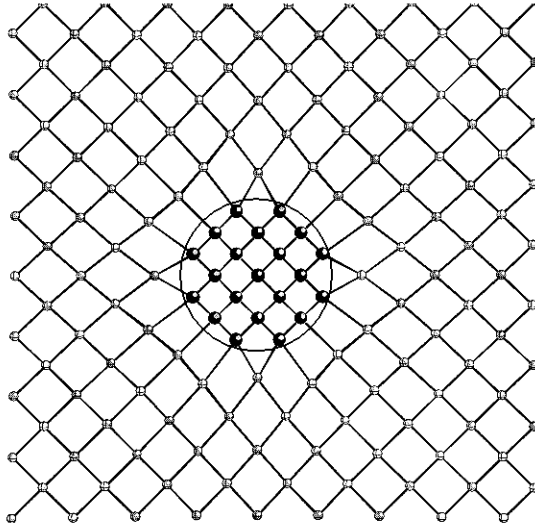
- Powerful single crystal techniques applied to polycrystals
- ~4-6 Orders of magnitude lower activity
 - Safer/lower backgrounds
- Cryocooled samples to study initial defects
- New information about point/line/mesoscale defect interactions



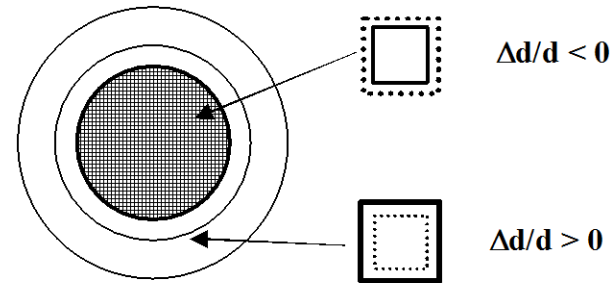
Successful demonstration experiments!

Vacancies, interstitials, small dislocation loops, coherent precipitates are additional type 1 defects

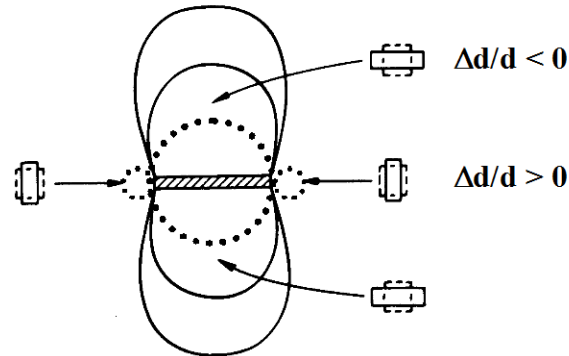
Lattice Distortions



Coherent Precipitate

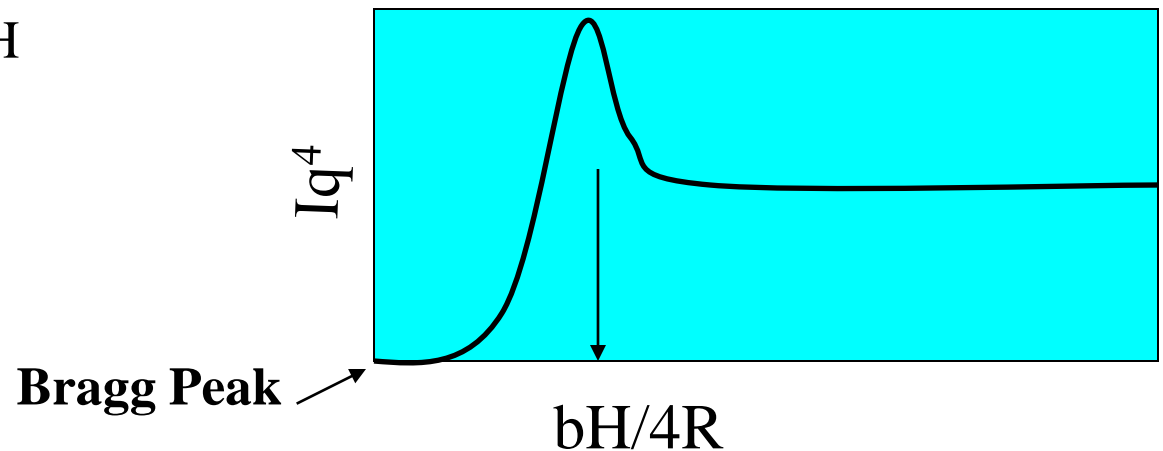
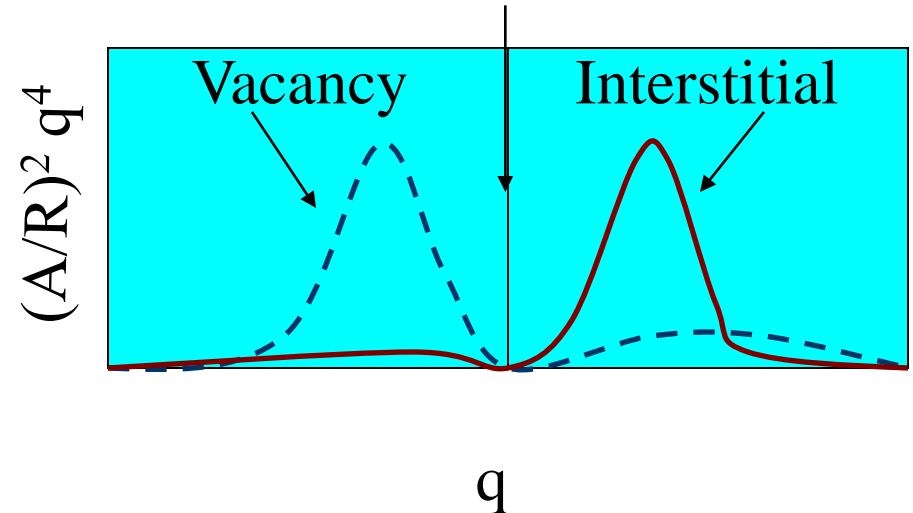


Dislocation Loop

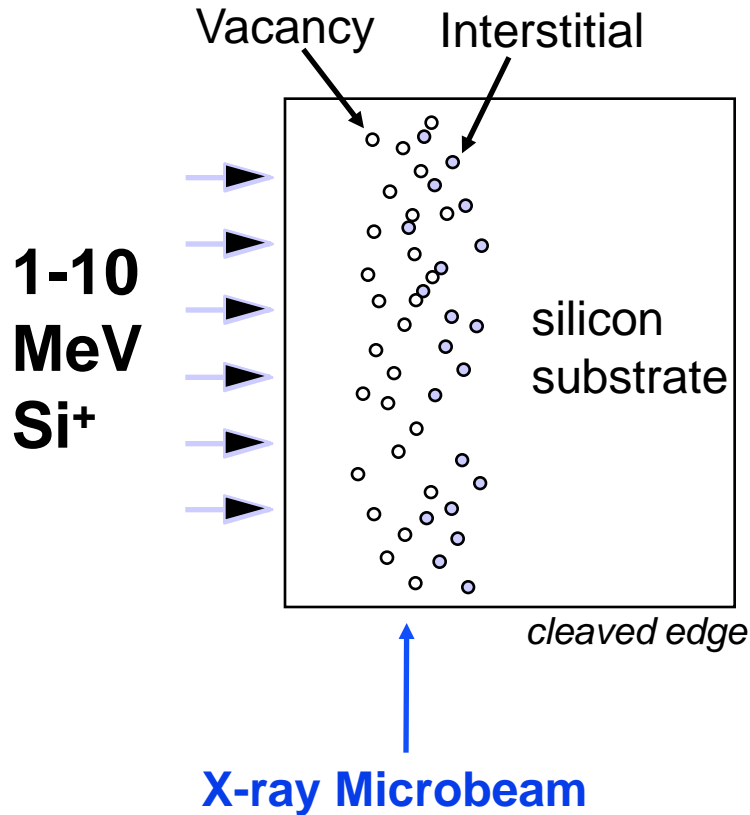


Numerical calculations determine quantitative cross sections

- Sign of diffuse scattering reverses for vacancy/interstitials
- For interstitial loops- enhanced scattering at $q=bH/4R$
- For coherent precipitates enhancement at $q=-\varepsilon H$

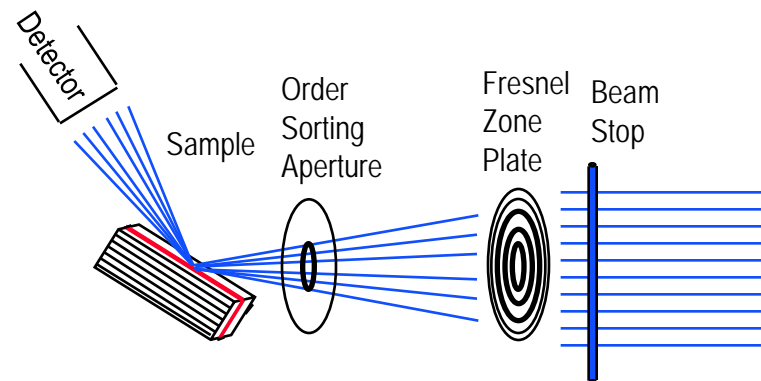
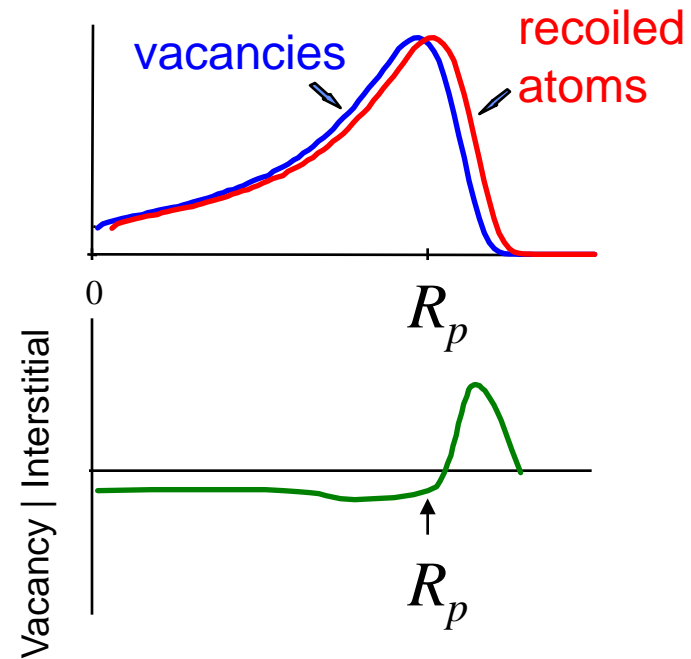


Micro-diffuse scattering applied to High Energy, Self-Ion Implantation in Si



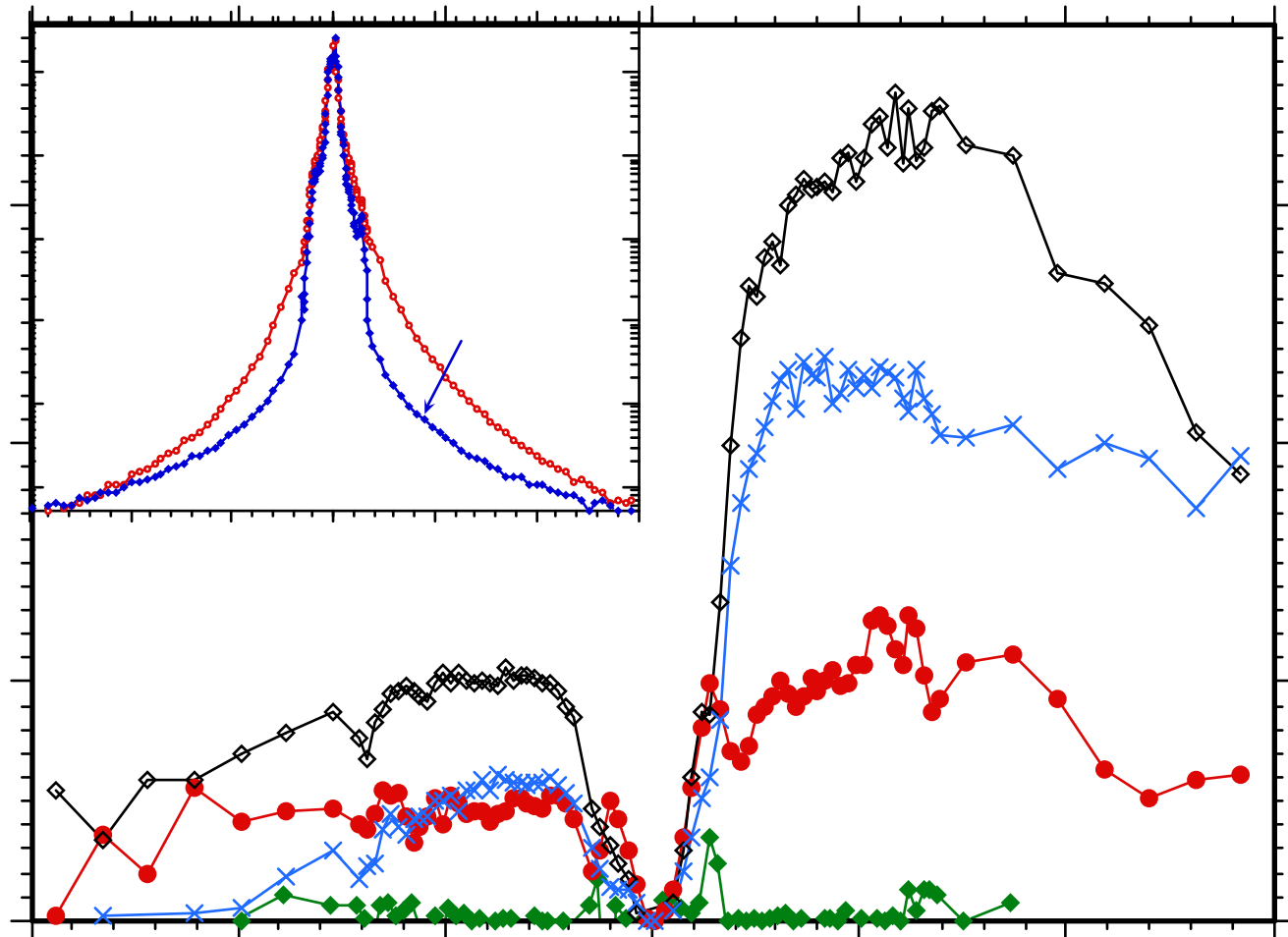
- cleave sample in cross-section
- translate to probe depth dependence

Spatial separation of recoils and vacancies due to momentum transfer

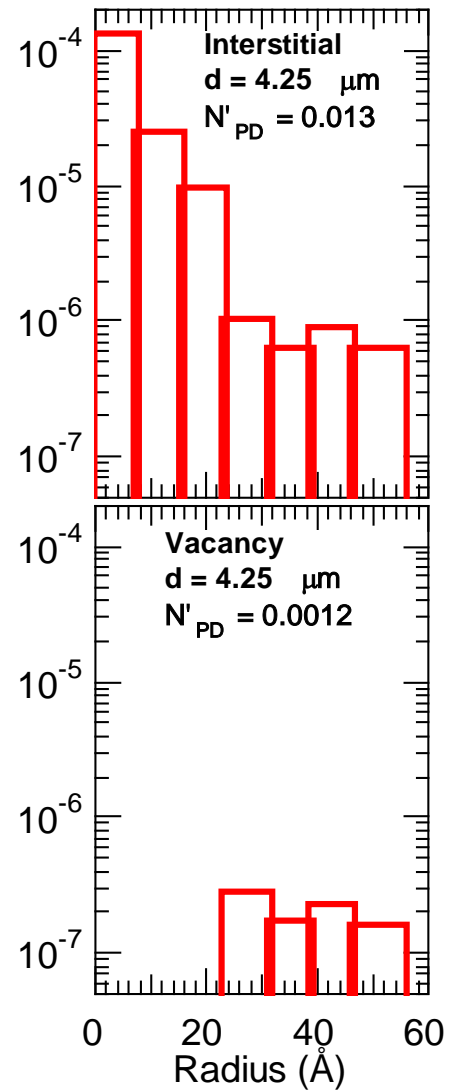
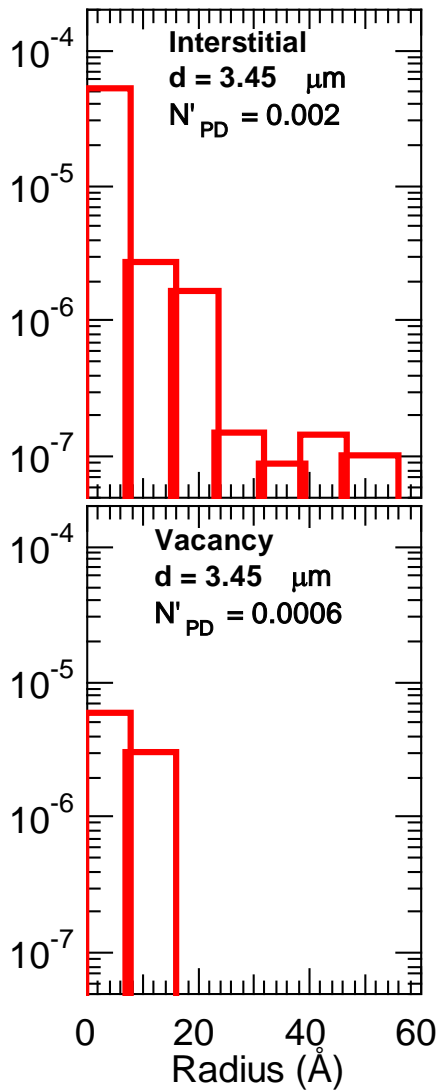
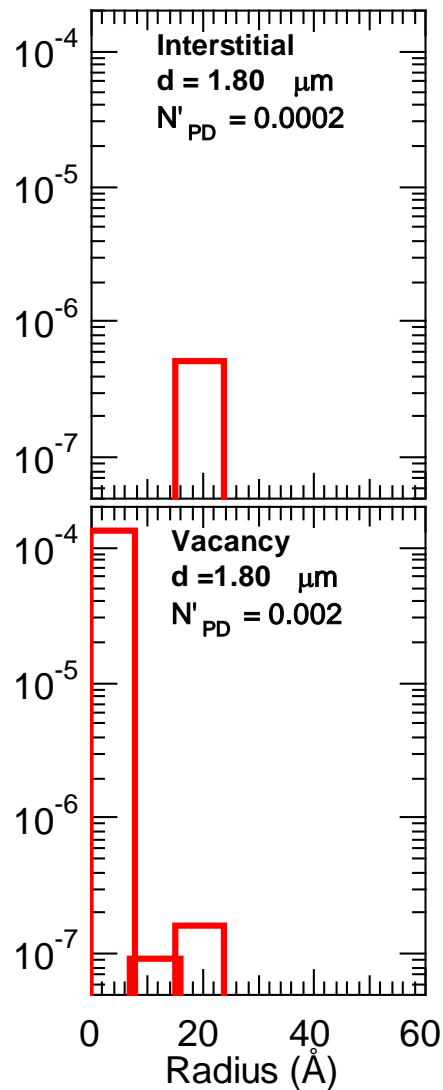


X-ray Diffuse Scattering

Huang theory \Rightarrow for $Q \ll 1/R$, $I \propto Kb\pi R^2/Q^4$

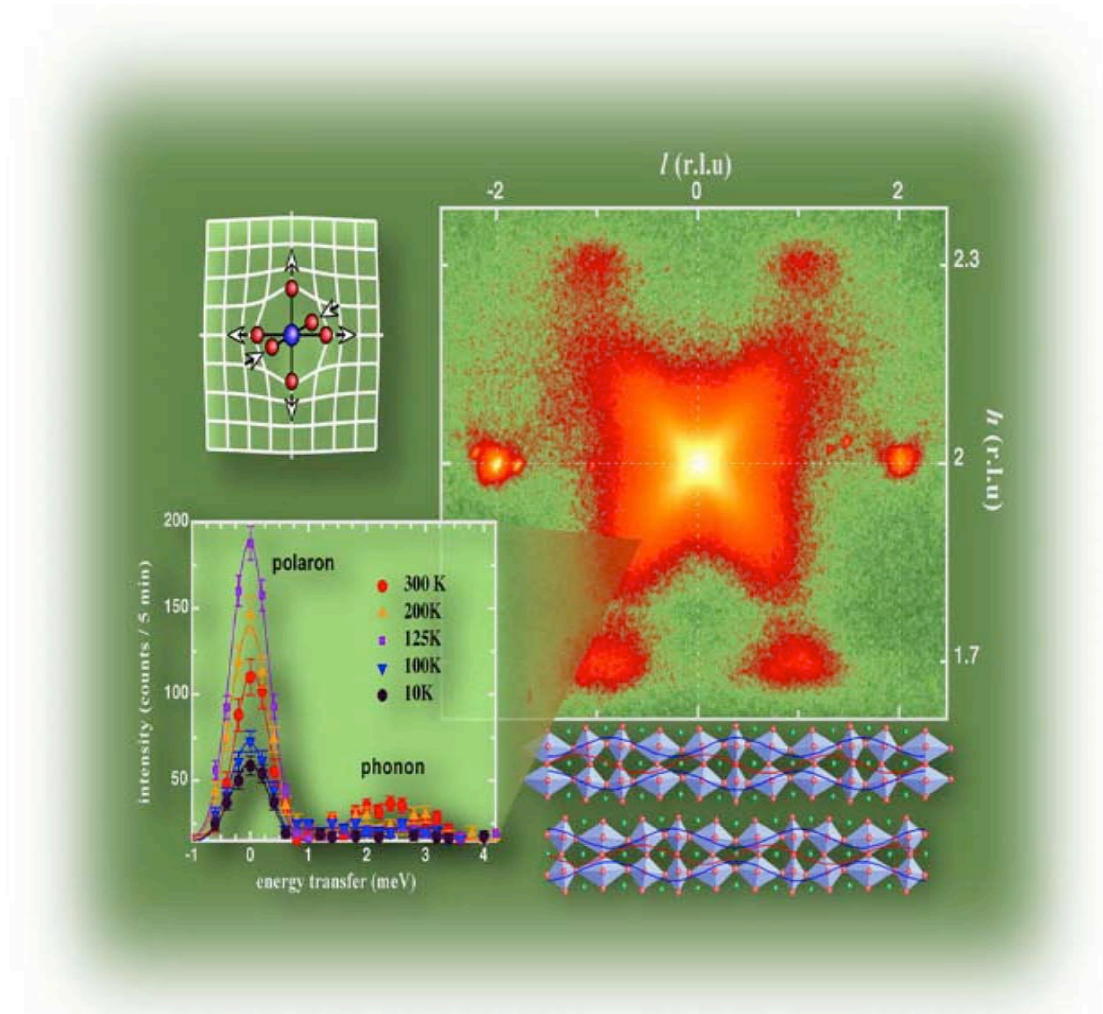
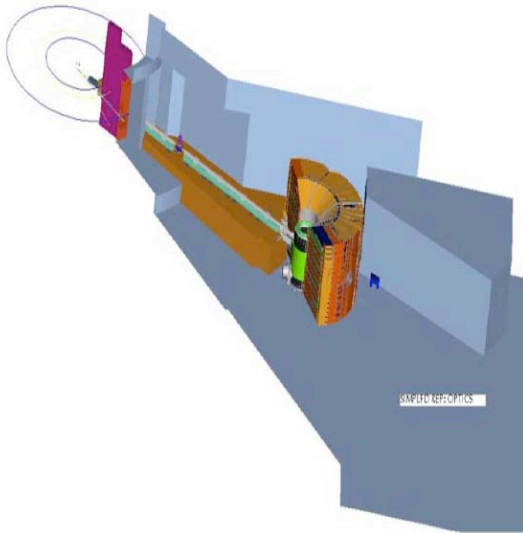


Depth Dependence of Size Distributions for Ion-Implanted Si



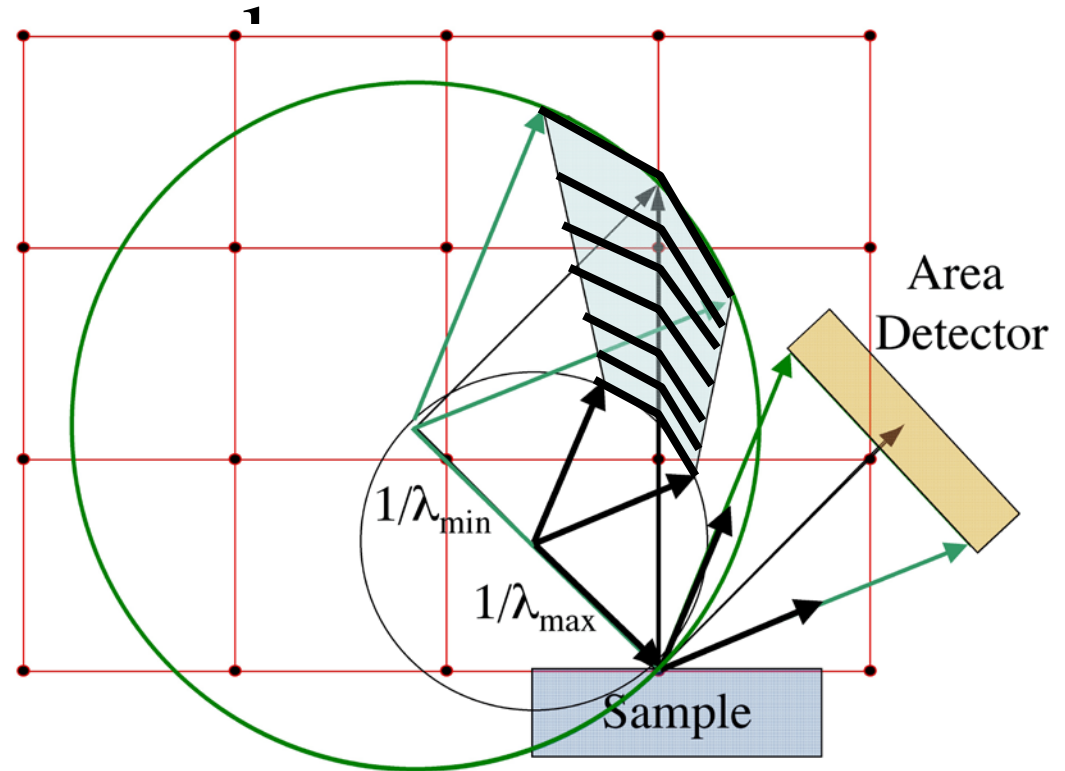
Corelli SNS beamline specialized for diffuse scattering with elastic Discrimination

- Complex disorder and short-range correlations



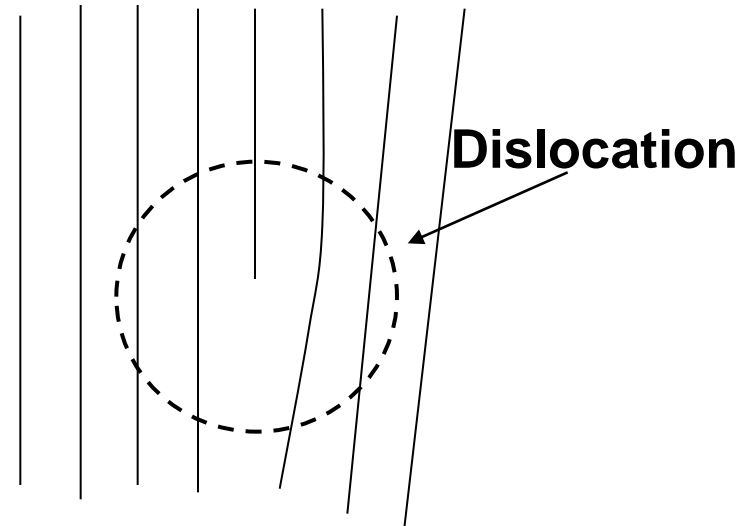
Nondispersive tunable microbeam will
measure diffuse scattering without moving

- Tunable beam moves $<0.5 \mu\text{m}$
- Area detector collects data in parallel
- Cooling needed to reduce background
- Must eliminate harmonics



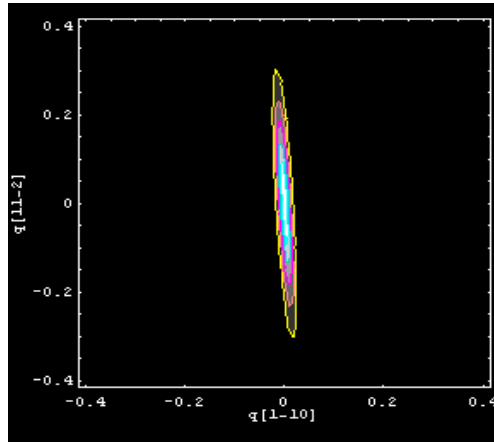
Dislocations examples of defects of the 2nd kind where microbeams already provide new information

- Long-range effect on lattice sites
- Paired dislocations causes broadening of Bragg peak
- Unpaired causes macroscopic rotations of crystal planes with streaking of reflections

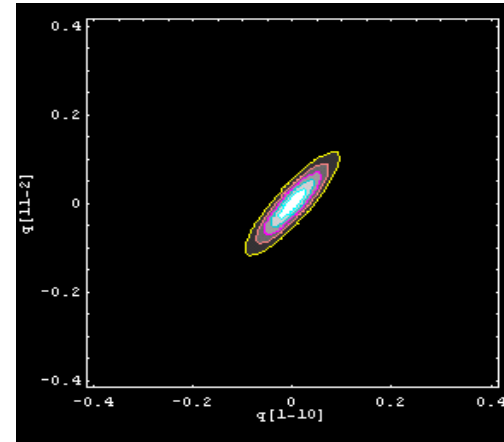


Influence of number and orientation of dislocations can be quantified

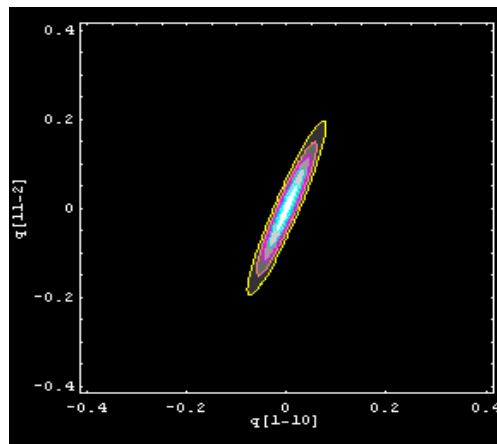
◆=[1-2-1], n=[-1-11], b=[101]



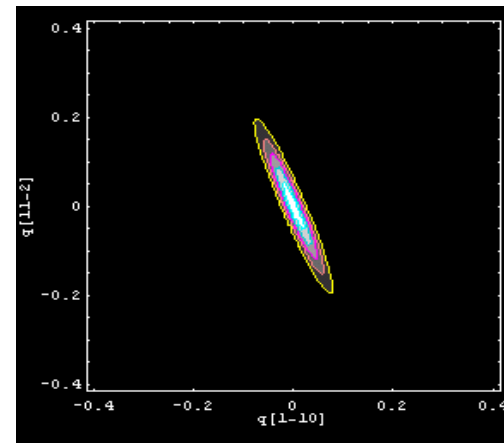
◆=[-1-21], n=[-111], b=[101]



◆=[-11-2], n=[1-1-1], b=[110]

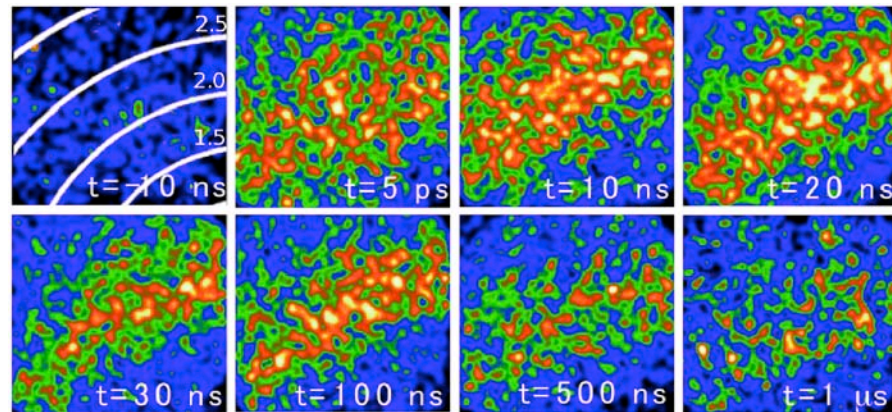


◆=[1-1-2], n=[1-11], b=[110]



X-ray diffuse scattering at Femtosecond Resolution

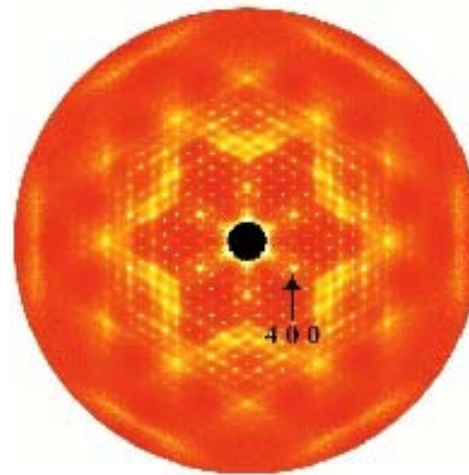
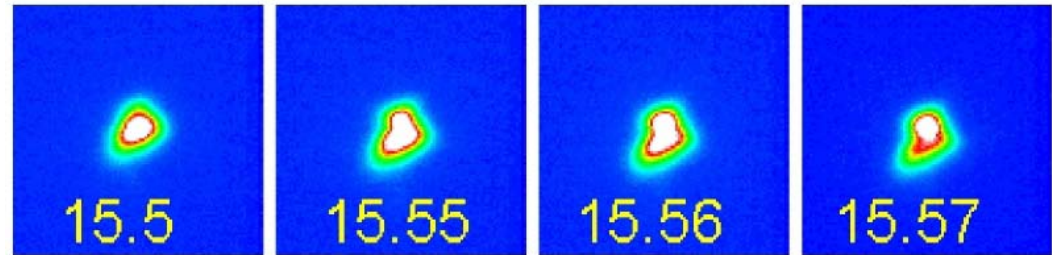
- Ultra-brilliant LCLS opens new experimental possibilities
- Transient behaviors at femtosecond time scales demonstrated.



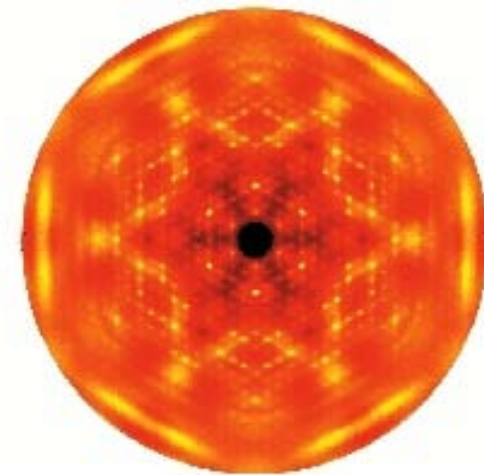
Lindenberg et al. PRL 100 135502 (2008)

New directions in diffuse scattering

- High-energy x-ray
- Microdiffuse x-ray scattering
 - Combinatorial
 - Easy sample preparation
- Diffuse neutron data from every sample
- Interpretation more closely tied to theory
 - Modeling of scattering x-ray/neutron intensity



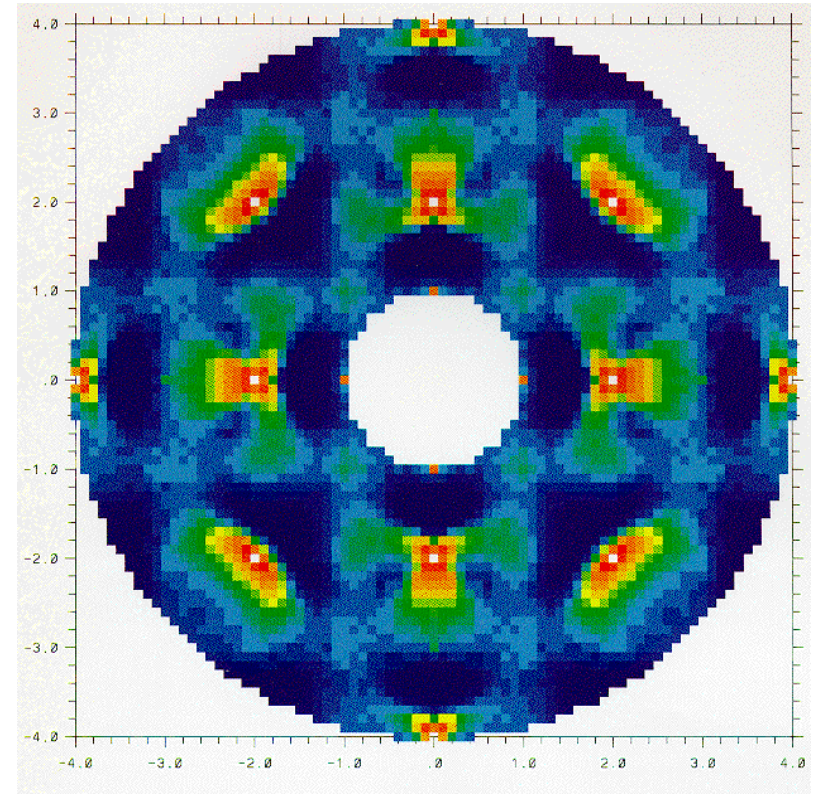
Experiment



Model

Intense synchrotron/neutron sources realize the promise envisioned by pioneers of diffuse x-ray scattering

- M. Born and T. Von Karman 1912-1946- *TDS*
- Andre Guiner (30's-40's)-*qualitative size*
- I. M. Lifshitz *J. Exp. Theoret. Phys. (USSR)* **8** 959 (1937)
- K. Huang *Proc. Roy. Soc.* **190A** 102 (1947)-*long ranged strain fields*
- J. M Cowley (1950) *J. Appl. Phys.*-*local atomic size*
- Warren, Averbach and Roberts *J. App. Phys* **22** 1493 (1954) -*SRO*
- Krivoglaz *JETP* **34** 139 1958 *chemical and spatial fluctuations*

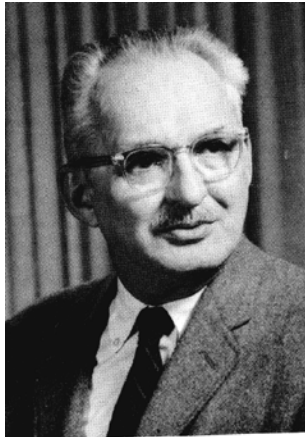


Other references:

- X-ray Diffraction- B.E. Warren Dover Publications New York 1990.
- http://www.uni-wuerzburg.de/mineralogie/crystal/teaching/dif_a.html
- Krivoglaz vol. I and Vol II.

Diffuse scattering done by small community

- Warren school



S. Cowley, Arizona St.

Bernie Borie, ORNL

Jerry Cohen, Northwestern

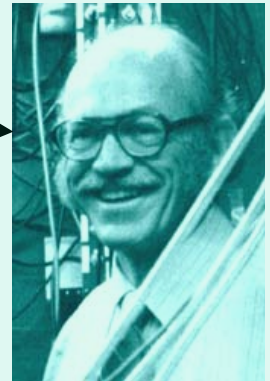
B. Schoenfeld, ETH Zurich

W. Schweika, KFA Jülich

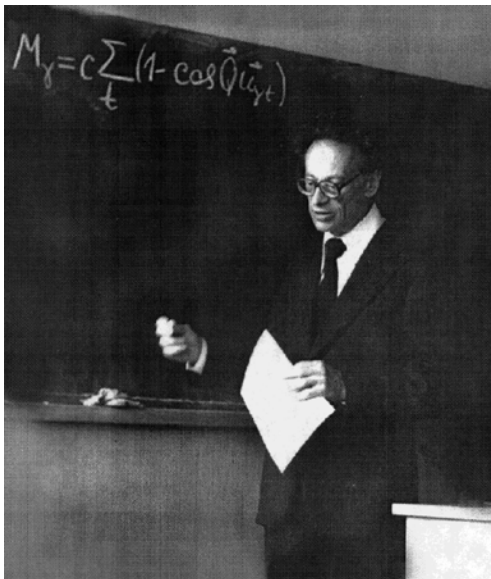
Simon Moss, U. Of Houston



Cullie Sparks
ORNL



- Krivoglaz school

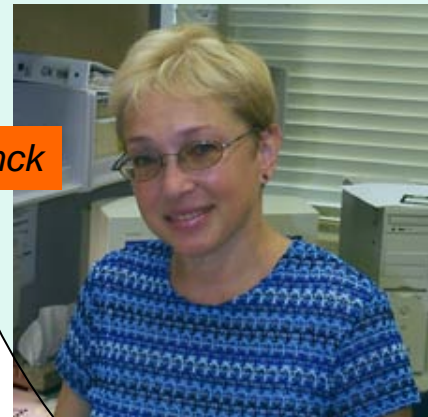


Peisl, U. München

H. Reichert, Max Planck

Gitgarts, Minsk

Rya Boshupka, IMP



Rosa Barabash
ORNL

Diffuse scattering song

Come eager young scholars- so tender and new
I'll teach you diffraction- what I says mostly true
Between the Bragg Peaks lies a world where you see
Fluctuations and defects- they stand out plane-ly

Chorus

For its dark as a dungeon between the Bragg peaks
But here in the darkness- each defect speaks
It gathers- from throughout- reciprocal space
And re-distributes all over the place.

Between the Bragg peaks - one thing that we see
Is TDS on our CCD
Intensity totals are conserved- you can't win
It steals from the Bragg peaks that stay very thin

Substitutional alloys can cause quite a stir
The shorter the length scale the greater the blur
With care you can find out the bond length between
Each atom pair type-the measurements clean

Dislocations and other- type 2 defects
Destroy the Bragg peaks -they turn them to wrecks
But near the Bragg peaks- you still can see
Intense diffraction continuously

Many -are- the defects you find
Between the Bragg peaks where others are blind
So go tell your friends and impress your boss
You've new understanding -with one hours loss

